

Advances in Surface Passivation of Nanoscale Zerovalent Iron A Critical Review

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1	Advances in surface passivation of nanoscale
2	zerovalent iron (NZVI): A critical review
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24 ABSTRACT

25 Nanoscale zerovalent iron (NZVI) is one of the most extensively studied nanomaterials in the fields of wastewater treatment and remediation of soil and groundwater. However, rapid 26 oxidative transformations of NZVI can result in reduced NZVI reactivity. Indeed, the surface 27 passivation of NZVI is considered one of the most challenging aspects in successfully 28 29 applying NZVI to contaminant degradation. The oxidation of NZVI can lead to the formation of Fe^{II}-bearing phases (e.g., Fe^{II}O, Fe^{II}(OH)₂, Fe^{II}Fe^{III}₂O₄) on the NZVI surface or complete 30 oxidation to ferric (oxyhydr)oxides (e.g., Fe^{III}OOH). This corrosion phenomenon is dependent 31 32 upon various factors including the composition of NZVI itself, the type and concentration of aqueous species, reaction time and oxic/anoxic environments. As such, the co-existence of 33 34 different Fe oxidation states on NZVI surfaces may also, in some instances, provide a unique reactive microenvironment to promote the adsorption of contaminants and their subsequent 35 transformation via redox reactions. Thus, an understanding of passivation chemistry, and its 36 related mechanisms, is essential not only for effective NZVI application but also for 37 38 accurately assessing the positive and negative effects of NZVI surface passivation. The aim of this review is to discuss the nature of the passivation processes that occur and the passivation 39 40 byproducts that form in various environments. In particular, the review presents: i) the strengths and limitations of state-of-the-art techniques (e.g., electron microscopies and X-ray 41 42 based spectroscopies) to identify passivation byproducts; ii) the passivation mechanisms 43 proposed to occur in anoxic and oxic environments; and iii) the effects arising from synthesis 44 procedures and the presence of inorganics/organics on the nature of the passivation 45 byproducts that form. In addition, several depassivation strategies that may assist in increasing and/or maintaining the reactivity of NZVI are considered, thereby enhancing the 46 47 effectiveness of NZVI in contaminant degradation.

48

49 **1. INTRODUCTION**

In recent decades, zerovalent iron (ZVI, Fe(0)) has been extensively studied for its 50 capacity to transform contaminants in soils, groundwater and industrial wastewaters because 51 of: (i) its ability to reductively transform a wide variety of compounds, such as halogenated 52 organics, into less toxic and/or more biodegradable forms; (ii) the abundance of iron (the 53 54 fourth most abundant element in the earth's crust); (iii) the relative environmental friendliness of an iron-based technology compared to other zerovalent metals (such as Mn(0) and Cu(0)) 55 and (iv) the low cost of Fe⁰ mass production.¹⁻⁷ In particular, nanoscale zerovalent iron (NZVI) 56 has attracted attention due to the potential advantages arising from high particle reactivity 57 (resulting from the small size and high surface area of NZVI) when compared to ZVI.^{3-6,8-21} 58 Moreover, NZVI has been confirmed to reductively remove various organic (e.g., chlorinated 59 aliphatic^{3,6,8-10} and aromatic compounds^{10,11,13,14}), inorganic (e.g., nitrate, ¹² Cr(VI), ¹⁷ As(III)¹⁸⁻ 60 ²² and Cd(II)²³) and radioactive (e.g., U(VI)²⁴⁻²⁶ and Tc(VII)²⁷) contaminants under anoxic 61 conditions. It has also been demonstrated that NZVI can induce the oxidative degradation of 62 organic contaminants (e.g., pharmaceuticals^{28,29} and herbicides^{30,31}). This may occur as a 63 result of the reduction of dissolved oxygen (O₂) and the associated production of hydrogen 64 peroxide (H_2O_2) that subsequently reacts with Fe(II) (that is also produced during oxidation) 65 leading to initiation of the Fenton reaction (Fe(II) + $H_2O_2 \rightarrow Fe(III) + HO^* + OH^-$) and the 66 generation of strongly oxidizing hydroxyl radicals (HO^{*}, 2.80 V). 67

Typical NZVI particles have a core-shell (Fe(0)-Fe (oxyhydr)oxide) structure with the spherical nanosized primary particles forming chain-like aggregates with sizes ranging up to a few micrometers.³ Moreover, the aggregates exhibit the magnetic properties of the primary particles with the magnetism of the primary particles influencing the size and structure of these assemblages.³² The co-existence of the Fe(0) core and oxidized Fe surface layer provides a unique reactive surface for the initial adsorption of contaminants and their

74 subsequent transformation on the particle surface via reductive or oxidative pathways.^{3,5,14,16,29,30} In accordance with contaminant adsorption, oxidation or reduction, Fe 75 may undergo oxidation (from Fe(0) to Fe(II) and Fe(II) to Fe(III)), precipitation (as Fe(II) 76 and/or Fe(III) solids) and, possibly, co-precipitation with ionic species.^{8,33–35} Concomitantly, 77 it may also be necessary to consider reactions with other solution phase entities such as O_2 78 (which, as noted above, may be reduced to H₂O₂, possibly due to adsorption to NZVI or its 79 (oxyhydr)oxide-coated surface),^{28,29} sulfate (which may possibly be reduced to sulfide, 80 resulting in the formation of FeS solid phases on the NZVI surface)³⁵ and carbonate (which 81 may also promote the precipitation of ferrous carbonate minerals).³⁶ 82

Many researchers have demonstrated that NZVI undergoes surface oxidation (or surface 83 passivation) when used for the removal of contaminants in variably oxic environments.^{3,6,8,29} 84 In addition, a field study has shown that 78–97% of Fe(0) injected was oxidized in a sediment 85 sample after 165 d at a remedial site contaminated with a dense non-aqueous phase liquid 86 (DNAPL).³⁷ Investigations have revealed that, depending on the environmental conditions 87 and types of contaminants present, NZVI surface layers (and eventually, the bulk of the solid) 88 transform to different Fe mineral phases, such as vivianite (Fe^{II}₃(PO₄)₂·8H₂O), Fe(OH)₂, 89 green rust (Fe(II)-Fe(III) layered double hydroxides with various interlayer anions (e.g. Cl⁻, 90 SO_4^{2-} and CO_3^{2-})), ferrihydrite, magnetite (Fe^{II}₁^{III}₂O₄), lepidocrocite (γ -Fe^{III}OOH), goethite (α -91 Fe^{III}OOH), mackinawite (Fe^{II}S), and siderite (Fe^{II}CO₃), as presented in Table 1. Moreover, 92 NZVI can be completely transformed to lepidocrocite in oxygenated water³⁸ and partially 93 transformed to Fe(OH)₂ in O₂-free water in the absence of contaminants.³⁹ The passivation 94 95 kinetics and the nature of the products generated have a significant influence on the long-term viability of the technology given that Fe(II)-containing minerals such as magnetite,⁴⁰⁻⁴³ 96 vivianite, $^{44-47}$ and green rust $^{48-50}$ and surface-bound Fe(II) 51,52 can further remove organic and 97 inorganic contaminants. In addition, Fe(III)-containing minerals (e.g., maghemite $(\gamma$ -Fe₂O₃),⁵³ 98

goethite⁵⁴ and hematite⁵⁵) can effectively remove contaminants via sorptive effects. For 99 example, it has been reported that, in anoxic conditions, passivated NZVI demonstrated 100 similar or slightly higher sorptive ability for perfluoroalkyl acids⁵⁶ and $As(V)^{57}$ when 101 compared to fresh NZVI. These results are most likely due to the presence of a greater 102 103 number of adsorption sites on the Fe (oxyhydr)oxides passivating the NZVI surface compared 104 to the number of sites on pristine NZVI. These findings imply that passivation byproducts of 105 NZVI can potentially act as reactants for the further removal of contaminants, even after the depletion of the initial NZVI. 106

107 Although the positive effects of passivation are evident, there are several adverse effects to the formation of passivation byproducts on NZVI that should be considered (in addition to 108 reduced activity). The production of reactive oxygen species (ROS) in NZVI suspensions, as 109 previously mentioned, was considered to account for the high toxicity of fresh NZVI to rodent 110 111 neuron and microglia cells when compared to the limited effect of aged NZVI and magnetite nanoparticles⁵⁸ with the severity of toxicity related to the iron oxidation state with Fe(0) >112 Fe(II) > Fe(III).⁵⁹ The toxicity of passivation byproducts from NZVI may be lower than that 113 of NZVI however the results of the above studies suggest that the passivation byproducts (i.e., 114 Fe(II) and Fe(III) species) can also induce secondary toxic effects. In addition, it has been 115 116 suggested that the presence of nanoparticles may enhance the transport of contaminants associated with these nanoparticles,⁶⁰ though certain reports suggest that the mobility of NZVI 117 particles is likely to be retarded as a result of the association of these particles with subsurface 118 sediments.^{61,62} Therefore, an understanding of passivation kinetics, transformation byproducts 119 120 and related mechanisms is required when considering the reactivity, longevity and stability of NZVI. 121

Although a number of reviews regarding NZVI have been published, most reviews havefocused on NZVI synthesis and characterization, application to the removal of contaminants

or implementation and results of field studies.^{15,63-74} In comparison, in this review, we 124 introduce: (i) common characterization techniques for determining the nature of passivation 125 byproducts; (ii) insights into the passivation mechanism(s) underlying observed structural 126 transformations of NZVI in oxic and anoxic environments and (iii) the effect of synthesis 127 methods and solution conditions such as the presence of various organic and inorganic species 128 129 on the passivation process. In addition, we describe various depassivation strategies that could 130 potentially be used to either prevent or retard passivation processes and thereby extend the 131 lifetime over which NZVI is an effective agent for the degradation of contaminants.

132

2. CHARACTERIZATION OF NZVI PASSIVATION BYPRODUCTS

To identify the NZVI passivation mechanism operating under specific conditions, the investigation and monitoring of the formation of passivation byproducts throughout and after the passivation process are required. In this section, common techniques used for identifying passivation byproducts are reviewed and the benefits and limitations of each technique are discussed. The primary methods of interest include electron microscopy, X-ray diffraction (XRD), X-ray photoelectron spectroscopy (XPS), Raman and Mössbauer spectroscopies and X-ray absorption spectroscopy (XAS).

141

142 2.1. Electron Microscopy

143 Transmission and scanning electron microscopy (TEM and SEM) have been the most 144 widely used techniques to determine the morphological characteristics of NZVI passivation 145 byproducts. TEM can provide visual information at the nanoscale level of properties such as the core size of fresh NZVI, the thickness of the passivating layer after partial passivation, and 146 147 morphological changes after complete structural transformation. Additionally, crystallographic information including crystallinity and lattice spacing can be obtained using 148

selected area electron diffraction.^{36,39} For fresh NZVI, a spherically-shaped core-shell 149 structure is commonly observed using TEM. The size of the core is typically within a range of 150 tens of nanometers with the core covered by a thin Fe (oxyhydr)oxide layer of several 151 nanometers thickness as shown in Figure S1.^{3,8,19,38,75} After surface passivation, 152 morphological changes in the NZVI structure can be detected using TEM. Typical changes 153 154 include loss of the Fe(0) core and the appearance of hollow (oxyhydr)oxide shells with this transformation reported to occur within 24 h under oxic conditions.³⁸ Further aging to 72 h 155 can result in the formation of flake- or acicular-shaped minerals (such as lepidocrocite).³⁸ 156

Another effective analytical technique typically associated with TEM is X-ray energy 157 dispersive spectroscopy (XEDS). The results of XEDS can provide the elemental composition 158 of passivation byproducts and can support identification of the Fe mineral phases present, 159 especially when certain elements such as P and S are co-localised in minerals such as 160 (respectively) vivianite (Fe₃(PO₄)₂) and mackinawite (FeS).^{33,34,35} In recent times, scanning 161 TEM (STEM)-XEDS has been used to identify the passivation mechanism that occurs in 162 single nanoparticles by the acquisition of nanoscale images and elemental maps.^{36,38,76} For 163 instance, the growth of amorphous assemblages within the NZVI core with subsequent 164 transformation to plate-like crystalline forms has been observed using STEM (Figures 1(a-c)). 165 166 These results can then be matched to XEDS maps to reveal a decrease and increase, respectively, in the Fe and O contents within the NZVI core.³⁶ 167

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169 **2.2. X-ray Diffraction (XRD)**

X-ray diffraction is one of the most popular analytical techniques for characterizing the
particular minerals that form upon NZVI passivation.¹⁵ It is particularly useful for identifying
crystalline phases that give sharp and characteristic diffraction peak patterns. Most studies in
which the mineralogy of passivated NZVI has been examined by XRD have typically been

limited to analyses 'before' and 'after' reaction of the sample with very limited insight gained 174 into the kinetics of mineral transformations.^{22,77–79} A large number of iron crystalline phases 175 have been identified by XRD in passivation byproduct studies however lepidocrocite and 176 magnetite/maghemite are overwhelmingly the dominant phases detected.¹⁵ The latter two 177 178 minerals cannot be identified separately by XRD as both have indistinguishable diffraction 179 patterns despite their different oxidation states. Other less common NZVI passivation byproducts detected by XRD include ferrihydrite,⁷⁷ goethite,⁸⁰ vivianite,^{33,34} amakinite 180 $(Fe^{II}(OH)_2)$,⁸¹ hematite,¹⁸ chukanovite $(Fe^{II}_2(OH)_2CO_3)^{82}$ and carbonate green rust.⁸³ 181

Perhaps the most significant limitation of using XRD to identify passivation byproducts 182 is that some minerals, including NZVI itself, quite often present as X-ray amorphous and/or 183 only display broad diffraction peaks.¹⁵ In addition, the mineral detection limits for XRD are in 184 the order of 2 to 3 wt. %.¹⁵ Thus, quantitative analysis of the relative proportions of minerals 185 present could be difficult or impossible with this analytical approach. This could possibly 186 explain why the very common nanocrystalline, X-ray amorphous mineral ferrihydrite has 187 rarely been identified as a major passivation byproduct in NZVI studies in which XRD was 188 used as the primary characterization technique. 189

190

191 2.3. X-ray Photoelectron Spectroscopy (XPS)

192 X-ray photoelectron spectroscopy is a surface sensitive technique that can be used to 193 characterize several nanometers of the NZVI outer layer. Although it is difficult to 194 conclusively determine the nature of the NZVI passivation byproducts using XPS due to the 195 limited information that can be obtained on the mineral phases, this technique can provide 196 important information on the nature of the elements present at the NZVI surface including 197 their oxidation states. The kinetic energy and number of electrons emitted from irradiation of 198 the surface by X-rays of known energy are measured during XPS analysis. For the

determination of the oxidation state of elements present within the surface layer, the narrow 199 peaks for the $Fe(2p_{3/2})$ and O(1s) binding energies are particularly useful.^{8,40,84} A 200 comprehensive understanding of the oxidation states of the Fe and O species on the NZVI 201 surface can provide critical information on passivation byproducts.^{38,84–86} For example, the 202 203 presence of an Fe(0) peak at \sim 706 eV indicates that the core Fe(0) is preserved because XPS is sensitive to the outermost layer of the particles.⁶ In addition, it has been reported that after 204 205 72 h reaction with oxygenated water, NZVI passivation byproducts displayed peaks of Fe at ~711 eV and O at ~530 eV and 531 eV, indicating the presence of Fe₂O₃, Fe₃O₄ and/or 206 FeOOH.³⁸ 207

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209 2.4. Raman and Mössbauer Spectroscopies

Raman spectroscopy can be carried out in situ to monitor the formation of NZVI 210 211 passivation byproducts, particularly when the minerals to be identified are poorly crystalline. However, the transformation of Fe (oxyhydr)oxides as a result of laser-induced thermal 212 heating may complicate interpretation of results.⁷⁷ For example, hematite is often detected by 213 Raman spectroscopy but absent in corresponding XRD data.^{77,78} In the early stages of NZVI 214 aqueous oxic corrosion, wustite (Fe^{II}O) can be detected by Raman spectroscopy (~595 cm⁻ 215 1)⁷⁸. It is an unstable intermediate phase generally formed in small amounts below the 216 217 detection limit of XRD. At longer aging times (e.g. 30 d in static deionized water or 24 h in 218 deionized water mixed at 300 rpm), Raman analyses have revealed the simultaneous coexistence of maghemite (665 and 730 cm⁻¹), magnetite (670 cm⁻¹) and lepidocrocite (650 cm⁻¹) 219 ¹).^{77,78} On prolongation of NZVI aging time to 60–90 d in static water or 48-72 h in stirred 220 water, only lepidocrocite (250, 380, 526 and 650 cm⁻¹) was identified (in close agreement 221 with XRD results).³⁸ Raman analysis has been used to show that aging under anoxic 222 conditions favors the formation of wustite, goethite and akaganeite (β-Fe^{III}OOH).⁷⁸ 223

⁵⁷Fe Mössbauer spectroscopy can be used for detecting Fe atoms in different oxidation 224 states and is a useful technique to identify the nature of any amorphous mineral phases present. 225 Mössbauer spectroscopy can provide relevant information on the valency and spin states of Fe 226 species as well as their magnetic properties via analysis of the hyperfine parameters. As such, 227 magnetite and maghemite can be distinguished using this technique. Moreover, it can be used 228 229 to determine the percentages of elemental Fe(0) and other secondary minerals that are present.⁶³ The typical ⁵⁷Fe Mössbauer spectrum of ZVI exhibits a single magnetic component 230 corresponding to Fe(0) with hyperfine parameters corresponding to a cubic phase of metallic 231 232 Fe. However a simple spectrum of this nature has rarely been observed for NZVI, even when the NZVI was pristine and freshly- prepared.^{19,63,87} Typically, a complex hyperfine structure is 233 observed with at least five components: a magnetic sextet with narrow lines corresponding to 234 that of Fe(0), two quadruple components corresponding to Fe(III) and Fe(II) species, and two 235 236 magnetic components. Mössbauer specialists suggest that these results are indicative of the presence of disordered ferric (oxyhydr)oxides (e.g., Fe(OH)₃) and/or superparamagnetic ferric 237 oxides (e.g., Fe_3O_4) on the passivated surface of Fe(0). When NZVI reacts with contaminants 238 in water, a quadruple doublet and a magnetic sextet with broadened lines are commonly 239 observed and have generally been attributed to the formation of lepidocrocite and a mixture of 240 magnetite and maghemite, respectively.^{19,87} 241

242

243 2.5. X-ray Absorption Spectroscopy (XAS)

Despite the widespread use in the fields of geochemistry and material science, XAS has had a limited application in NZVI studies with only a handful of studies describing the use of this technique to characterization of NZVI^{15,33,88,89} or NZVI passivation byproducts.^{36,79,86,90–92} One of the great strengths of Fe K-edge XAS is that it can provide quantitative insight into all the iron minerals present. This is because the high energy of the Fe K-edge (7112 eV for

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Fe(0)), allows for the entirety of the particles to be sampled by the beam. As an XAS spectrum is a linear combination of its components, it can then be fitted with reference material spectra in a linear least-squares fitting manner using software such as ATHENA.⁹³ As recommended elsewhere,⁶⁷ the choice of reference material spectra and the number of components to fit sample spectra should be guided on evidence from complementary analytical techniques, knowledge of the NZVI system, or principle component analysis and target analysis.

In the NZVI studies mentioned above, either the X-ray absorption near-edge structure 256 (XANES) or extended X-ray absorption fine structure (EXAFS) regions of the XAS spectrum 257 have been modelled. Generally, the Fe K-edge XANES region can be used for fitting spectra 258 when there are multiple Fe oxidation states present.^{86,89–91} However, it is a less useful region 259 to distinguish between Fe(III) oxyhydroxide passivation byproducts. For example, the 260 XANES spectra of ferrihydrite, lepidocrocite and goethite can, for fitting purposes, be 261 practically identical when using a Si(111) monochromator. If the identity of the exact Fe(III) 262 oxyhydroxides is an important component of a NZVI study, then the EXAFS region is better 263 suited for fitting of data.^{36,92} Interestingly, all the XAS studies that have examined NZVI 264 passivation byproducts have done so under oxidative conditions. Nevertheless, as noted in 265 266 many other NZVI studies, lepidocrocite, magnetite and maghemite are the main secondary iron minerals that have been detected by XAS analysis^{36,79,90-92} though ferrihydrite and 267 goethite have also been reported.^{36,79,92} 268

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- 270

0 3. NZVI PASSIVATION PROCESSES UNDER ANOXIC AND OXIC CONDITIONS

In this section, we review the state of knowledge with regard to the mechanism of passivation of NZVI under both anoxic and oxic conditions. Under anoxic conditions, the reaction with water is expected to be of principal importance for the formation of passivating layers on the NZVI surface though attention is also given to the impact of other major
solution constituents (such as carbonate, sulfate and nitrate) that may be present. Under oxic
conditions, the presence of oxygen is obviously the major differentiator to anoxic systems
with O₂ potentially influencing the reaction pathway of Fe(II) resulting from NZVI corrosion.
Consideration is also given to the impact of the nature of the NZVI particle on the oxic
passivation process with bimetallic particles of major interest.

280

281 **3.1. Anoxic Conditions**

With respect to the anoxic reaction of NZVI with water, the identification of the 282 primary reductive products (particularly Fe(OH)₂ and magnetite) and the definition of the 283 reaction mechanism have been well addressed by Filip et al.³⁹ These investigators³⁹ 284 used "NANOFER 25P" NZVI particles of ~60 nm in diameter and a specific surface area of 285 \sim 36 m²g⁻¹ that were thermally synthesized from iron oxide powder in H₂. Aging studies were 286 undertaken under strictly anoxic conditions at both 25 °C and 80 °C. ⁵⁷Fe Mossbauer 287 spectroscopy indicated that approximately 70% of the Fe(0) was transformed to Fe(OH)₂ after 288 ~3000 h of reaction at 25 °C with this mineral transforming relatively rapidly (after ~200 h) to 289 magnetite (Fe₃O₄) at 80 °C (Figure 2). Interestingly, the results obtained by Filip et al.³⁹ were 290 at odds with the results of earlier workers such as Reardon et al.⁹⁴ who concluded that 291 magnetite was also formed at 25 °C. Filip et al.³⁹ investigated this discrepancy and showed 292 293 that the nanosheets of Fe(OH)₂ formed in their study could be rapidly transformed to magnetite on exposure to air during analysis by XRD - the solid characterization method used 294 by Reardon et al.⁹⁴. Quantum chemical calculations indicated a two-step reaction mechanism 295 involving two one electron transfer processes; the first, which involved the breakage of the H-296 O bond of water, can be described by the reaction (Eq.1): 297

299	$\mathrm{Fe}^{0} + \mathrm{H}_{2}\mathrm{O} \rightarrow \mathrm{HFe}^{\mathrm{I}}\mathrm{OH}$	(1)
300		
301	which represents the rate-limiting step, and the second by (Eq.2):	
302		
303	$HFe^{I}OH + H_{2}O \rightarrow Fe^{II}(OH)_{2} + H_{2}$	(2)
304		
305	Filip et al. ³⁹ note that the existence of the HFeOH molecule has previously been demonstra	ated
306	(by Zhang et al. ⁹⁵) by argon matrix isolation FTIR spectroscopy.	
307	The nature of the corrosion products formed on NZVI upon exposure to anoxic solution	ions
308	has been investigated in a more recent study with Liu et al. ⁷⁸ concluding that the m	ajor
309	passivation byproducts formed after 72 h of reaction were wüstite (Fe ^{II} O), goethite	and
310	akageneite (Figure 3). These products differ markedly from those reported by Filip et al. ³⁹	and
311	beg the question of whether air exposure during TEM, XRD and Raman analyses resulted	d in
312	transformation of the passivation products formed under anoxic conditions. Pullin et a	al. ⁷⁹
313	investigated the nature of the passivation byproducts formed on exposure of NZVI to Mill	li-Q
314	water for periods of more than four weeks with the nature of the passivation byproduced	ucts
315	investigated by XAS. Amorphous passivation products (assigned as ferrihydrite) were for	med
316	within one day of exposure with the concurrent formation of magnetite. Over periods of u	p to
317	four weeks, maghemite was observed to form with this oxide transforming slowly to "m	eta-
318	stable" lepidocrocite. Over longer periods (greater than four weeks), all NZVI transforme	d to
319	goethite with this iron oxide considered to be the likely "final state"	
320	In summary, the anoxic corrosion of NZVI by water is dependent upon various fact	ors,
321	such as reaction time and temperature, with the major corrosion products observed be	eing
322	wustite, goethite and magnetite. It should be recognized, however, that the intrusion	n of
323	oxygen either through the aging process or during analysis may confound the results	s of

experiments undertaken under supposed anoxic conditions. As such, investigators should pay particular attention to the possibility of unintended NZVI passivation during studies under anoxic conditions.

327

328 **3.2.** Oxic Conditions

The nature of the oxidation products formed on exposure of nanometer-sized Fe(0)329 330 particles to oxygen has been of interest for many years in view of the widespread use of ultrafine magnetic particles in magnetic recording, ferrofluids and catalysis with the nature of 331 332 the oxide phases formed on exposure of small iron particles to an oxygen-containing atmosphere investigated by a number of groups.⁹⁶ Linderoth et al.⁹⁶ found, using Mössbauer 333 spectroscopy, that the oxide layer grows quickly initially then slows down with the oxide 334 layer consisting of a mixture of Fe₃O₄ and γ -Fe₂O₃. The kinetics of oxide layer formation 335 followed a logarithmic rate law in accord with the Caberra-Mott model of oxidation⁹⁷ with 336 multistage steps including i) dissociative adsorption of O₂ at the metal surface, ii) oxidation of 337 surface metal ions, iii) incorporation of metal ions into the oxide layer, iv) diffusion of metal 338 ions in the oxide layer, and v) reaction of metal ions with oxygen ions with the diffusion of 339 metal ions in the oxide layer considered to be rate limiting.⁹⁸ 340

In comparison with the logarithmic behavior of oxide layer growth revealed from 341 studies of passivation of iron particles in air, investigations of the oxidation of Fe(0) surfaces 342 in the presence of water vapour reveal that reactions at the iron surface/water interface are rate 343 limiting with an initial lag in oxide layer formation followed by an exponential increase in 344 growth of the oxide layer.^{99–101} Investigation of the aging of iron nanoparticles in aqueous 345 solution by Sarathy et al.¹⁰² revealed similar behavior with Fe(0) nanoparticles from Toda 346 347 (RNIP-10DS) becoming more reactive (as determined by the rate of carbon tetrachloride 348 (CCl₄) reduction) between 0 and ~ 2 days after exposure to water then gradually losing

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reactivity over the ensuing 100 days. These changes in reactivity were correlated with evidence for destruction of an Fe(III) oxide film that initially covered the Fe(0) nanoparticles with subsequent formation of a new passivating mixed valence Fe(II)-Fe(III) oxide shell.¹⁰²

Using guartz crystal microbalance measurements to quantify the rate of oxide layer 352 formation, Greenlee et al.¹⁰³ showed similar behavior to that observed by Sarathy et al.¹⁰² with 353 354 an initial lag in oxide layer formation following by an exponential increase in mass of this 355 layer (Region 1 in Figure S2). Decrease in water content of the solutions used (by replacing 356 water with ethanol) resulted in an increase in the lag period before oxidation commenced and 357 a decrease in the rate once oxidation had begun with these results confirming the critical role of H_2O in destabilizing the passivating layer initially present. As shown in Figure S2, this 358 exponential increase in oxide layer growth was followed by a logarithmic decline in rate of 359 corrosion at later times (Region 2) with this behavior in accord with the Caberra-Mott model 360 described above. Greenlee et al.¹⁰³ showed that NZVI oxidized primarily to lepidocrocite with 361 362 both carboxymethylcellulose (CMC) stabilizer and Ni slowing down the onset and extent of NZVI oxidation. Dong et al. ^{104–106} also showed that CMC and Ni slowed the aging of NZVI 363 with the major products of oxidation found to be magnetite and maghemite. 364

Recent studies of NZVI-mediated oxidation by He et al.³⁶ confirmed the results of 365 earlier studies by Joo et al.³⁰ that both Fe(II) and H₂O₂ are generated following 1 h exposure 366 of NZVI to oxygenated water at circumneutral pH. The solid Fe species resulting from 367 exposure of NZVI to oxygen were examined, as a function of exposure time, by EXAFS 368 spectroscopy.³⁶ Linear combination fitting of the EXAFS results indicated that NZVI 369 370 transformed to lepidocrocite via an intermediate ferrihydrite phase following exposure of NZVI to oxygenated water with essentially all NZVI transformed to ferric oxyhydroxides 371 after two hours of exposure to the oxic aqueous environment (Figure 1d).³⁶ 372

In summary, crystalline iron (oxyhydr)oxides such as lepidocrocite, maghemite and magnetite appear to be the major products formed upon the oxic corrosion of NZVI. Ferrihydrite may form initially but Fe(II)-catalyzed transformation to more crystalline forms is likely to occur as a result of the high concentrations of surface-associated Fe(II) present. The particular iron (oxyhydr)oxide formed is likely to be a function of the molar ratio of Fe(II) to Fe(III) surface sites present with the relatively short lifetime of Fe(II) in these oxic systems also influencing the extent of crystallization.

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381 4. FACTORS INFLUENCING NZVI PASSIVATION

Most of the published literature to date has addressed the impact of synthesis 382 methods¹⁰⁷ and both organic and inorganic compounds^{4,33,80,108–117} on the reactivity of NZVI 383 with respect to target contaminants in the context of groundwater or wastewater remediation. 384 385 In these studies, variations in the kinetic rate constants or extents of degradation were used as a means of assessing reactivity changes upon exposure to different cations or anions. 386 However, very few attempts have been made to determine the corrosion mechanism(s) and to 387 relate reactivity changes to the nature of the passivation byproducts formed¹⁰⁷ and the surfaces 388 exposed to co-occurring solutes.^{33,80} 389

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391 4.1. Effect of Synthesis Methods

The elemental composition of NZVI can vary depending on the synthesis method used,^{117,118} which may influence NZVI reactivity as well as the nature of the passivation byproducts. In most cases, Fe reduction is mediated by NaBH₄ (NZVI^{BH}) or H₂ (NZVI^{H2}) (Table 1). The XPS results revealed that the NZVI^{H2} surfaces composed of Fe (37.9–50.9%), O (44.2–52.9%), B (0%), Na (0.7–8.1%) and S (0.9–1.9%, possibly from the FeSO₄ precursor).¹⁰⁷ The surface of the NZVI^{BH} particles contained less Fe (20.0%–30.8%) and S

(0.0%-0.5%), but greater O (49.1-59.2%), B (12.0-16.0%) and Na (0.5-14.5%).^{107,118} 398 Moreover, it was found that the S on the NZVI^{BH} surface was in an oxidized form, whereas 399 the S present on the surface of the NZVI^{H2} particles was in reduced form.¹⁰⁷ Interestingly, 400 NZVI^{H2} exhibited lower reactivity than NZVI^{BH} for the transformation of CCl₄ to chloroform 401 402 (CHCl₃), presumably due to the high Fe/O ratio (possibly associated with the presence of a magnetite shell) on the NZVI^{H2} surfaces.¹⁰⁷ This hypothesis was supported by the low ability 403 of nanoscale magnetite to transform CCl₄.¹¹⁹ These results serve to demonstrate that synthesis 404 methods could alter the surface reactivity of NZVI and potentially its passivation mechanism. 405 As noted above, the use of NaBH₄ during the synthesis of NZVI^{BH} can result in the 406 occurrence of B on the surface of NZVI^{BH 120}. Recently, it has been reported that NaBH₄ can 407 induce the disintegration of NZVI^{BH} particles, resulting in the formation of significantly 408 smaller particles.¹²¹ This phenomenon may be induced by chemical etching of the passivated 409 NZVI surface (coated, for example, by iron oxides such as magnetite) with this result 410 consistent with the ability of NaBH₄ to induce the disintegration of microscale magnetite into 411 nanoscale magnetite.¹²² Although these authors did not compare their results to NZVI^{H2}, the 412 disparity in elemental surface composition (e.g., more Fe and S with less Na on the NZVI^{H2} 413 surfaces) may possibly cause reactions to proceed differently. 414

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4.2. Effect of Inorganic Compounds

Although the corrosion rate of NZVI and the nature of the passivation byproducts can be significantly affected by the presence of anionic or cationic inorganic species, very few studies have focused on this topic.^{33,80} The corrosion phenomenon is dependent on several factors including the type and concentration of aqueous species, reaction time and oxic/anoxic environments.^{33,80} It has been reported that the presence of common groundwater anions may promote or inhibit the corrosion of NZVI, however limited insight into the mechanism(s) of

the effects of anions is available.^{33,80} After six months of aging in pH unbuffered and anoxic 423 solutions containing 10 mM of either HCO₃⁻, SO₄²⁻, NO₃⁻, HPO₄²⁻ or Cl⁻, NZVI (commercial 424 sample, RNIP 10-DP) was found to predominantly transform to vivianite in the phosphate-425 aged sample and schwertmannite in the sulfate-aged sample (Figure S3).³³ Both the fresh 426 427 NZVI and the sample aged for six months in 10 mM nitrate predominantly contained Fe(0)and magnetite.³³ The pH of the suspensions containing the different anions ranged from 9.0-428 11.8, indicating that the observed transformations occurred under alkaline conditions.³³ Using 429 the same concentrations of the anionic species, NZVI^{H2} particles have also been found to 430 undergo corrosion into different iron phases after 16 weeks.⁸⁰ For example, HCO₃⁻ produced 431 goethite, NO_3^- resulted in the formation of magnetite/maghemite and SO_4^{2-} and CI^- produced 432 a mixture of magnetite/maghemite, lepidocrocite and goethite (Figure S4).⁸⁰ It was observed 433 that the corrosion rate of NZVI increased in the order $HCO_3^- > CI^-/SO_4^{2-} > NO_3^-$, however 434 pH was only measured in experiments in which the uptake of Cu and Zn by the corroded 435 NZVI was examined over a time period of 4 weeks.⁸⁰ As such, the influence of this critical 436 parameter on the nature of the formed passivation byproducts is unknown. Elsewhere, Kim et 437 al.¹¹⁷ have reported that a commercial NZVI (NANOFER 25DS) coated by S-based inorganic 438 compounds was oxidized to magnetite in a synthetic groundwater containing Cl⁻ (1.95 mN), 439 HCO₃⁻ (1.14 mN), NO₃⁻ (0.15 mN) and SO₄²⁻ (0.44 mM). In contrast, iron carbonate hydroxide 440 hydrate was detected as a major passivation byproduct in the presence of humic acid (2.5 mg 441 L^{-1}). 442

443 Xie and Cwiertny⁴ have reported that the presence of $C\Gamma$, SO_4^{2-} and CIO_4^{-} almost 444 completely inhibited the reduction of 1,1,1,2-tetrachloroethane by NZVI over a time period of 445 one month, irrespective of the nature or concentration of the anion present or pH. In contrast, 446 the longevity toward degrading this contaminant was dependent upon the concentrations of 447 NO_3^{-} and HCO_3^{-} with complete reactivity loss over 1 d and 14 d, respectively, in 25 mM

suspensions. In addition, XRD analyses suggested that the loss in reactivity in NO₃-448 suspensions resulted from Fe(0) conversion into magnetite, whereas the formation of iron 449 carbonate hydroxide limited reactivity in the HCO₃⁻ suspensions (Figure 4). Furthermore, Liu 450 et al.¹⁰⁸ reported that trichloroethylene (TCE) reduction rates by NZVI are highly dependent 451 452 on anion concentrations. For example, concentrations below 1 mM NO₃⁻ slightly increased the 453 reduction rate but the reaction was retarded at NO₃⁻ concentrations between 1 and 3 mM and ceased at 5 mM. It was speculated that high NO₃⁻ concentrations may shift the TCE reduction 454 reaction from one of cathodic control (i.e., the reduction of TCE) to anodic control (i.e., the 455 release of Fe^{2+} and electrons) resulting in significant surface passivation. 456

On the other hand, Su et al.¹⁰⁹ reported that the rate of degradation of hexachlorobenzene 457 by NZVI was not influenced by the presence of HCO_3^- , Mg^{2+} or Na^+ (over the concentration 458 range of 0.8–7.7 mM). However, the rate of degradation was enhanced in the presence of 7.7 459 mM Cl^{-} or SO_4^{2-} due to the promotion of corrosion but inhibited in the presence of 7.7 mM of 460 NO_3^- due to the competition between the anions and contaminants for reactive surface sites. 461 However, it has been observed that the addition of Na₂SO₄ inhibited the degradation of 462 pentachlorophenol (PCP) by Pd/Fe bimetallic nanoparticles in the range of 2.5-10 mM¹¹⁰ and 463 the decolorization of methyl orange by NZVI particles at 14 mM.¹¹¹ Researchers generally 464 465 attribute the inhibitory effects of common groundwater anions to their affinity for surface sites on the iron (oxyhydr)oxides coating the NZVI with the resulting surface complexes limiting 466 467 the ability of contaminants to associate with the surface. Consistently, the inhibition effect has been found to increase in the order of $Cl^- < SO_4^{2-} < HPO_4^{2}$ for nitrate reduction by Peerless 468 $Fe(0)^{112}$ and $Cl^{-} < SO_4^{2-} < HCO_3^{-} < HPO_4^{2-}$ for TCE reduction by NZVI.¹⁰⁸ 469

Given that the standard reduction potential of Na⁺ ions (-2.71 V) is below that of Fe²⁺ ions (-0.44 V), Na⁺ ions do not affect degradation reactions mediated by Fe(0).¹⁰⁹ However, Cu²⁺ (0.8–7.7 mM) has been shown to enhance the degradation kinetics of hexachlorobenzene

by NZVI whereas Fe^{2+} (0.8–7.7 mM) has been reported to inhibit the degradation reaction due 473 to the formation of passivation layers on the particle surfaces.¹⁰⁹ Moreover, Cu²⁺ (5–10 mM), 474 Ni^{2+} (5–10 mM) and Fe^{3+} (1.67–5 mM) have been observed to enhance PCP degradation by 475 Pd/Fe nanoparticles due to the deposition of reduced copper and nickel species on the Fe 476 surfaces.¹¹⁰ The standard reduction potentials of the Cu^{2+} (+0.34 V) and Ni²⁺ (-0.257 V) 477 cations are significantly higher than that of Fe^{2+} (-0.44 V) and, as such, the reduction of Cu²⁺ 478 or Ni²⁺ ions to their elemental forms is thermodynamically favorable.^{113,114} In the case of the 479 Fe³⁺ cation, the enhancement observed in the PCP degradation was attributed to a decrease of 480 pH resulting from the addition of the iron salt.¹¹⁰ Furthermore, spectroscopic investigations of 481 the structural evolution of NZVI in anoxic Co^{2+} solutions revealed that Co^{2+} was 482 heterogeneously reduced ($E_0 = -0.28$ V) and migrated to the NZVI core leading to its 483 continuous removal from the solution.¹¹⁵ However, information on whether Co²⁺ affects the 484 nature of passivation byproducts is lacking though dissolution of sheet or shell structures upon 485 metal complexation on the Fe surface has been reported in depassivation studies of micro-486 scale ZVI (~70 mesh) by Liu et al. (Figure 5).^{85,116} 487

In summary, anionic ligands such as phosphate, silicate and sulfate may be readily adsorbed onto iron corrosion products, affecting the reactivity of NZVI particles. Hard cations (such as Ca^{2+} and Mg^{2+}) and transition cationic metals (such as Ni^{2+} , Cu^{2+} and Co^{2+}) may exhibit promotive or inhibitive effects depending on their properties. However, the main secondary iron minerals formed during the reduction reaction have rarely been identified.

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494 4.3. Effect of Organic Compounds

The use of organic ligands in NZVI studies has been investigated in the context of colloidal stabilization to increase particle mobility and improve redox reactivity. For example, chemical stabilizers (such as humic acid, polyacrylic acid and anionic or nonionic surfactants) have been used to increase electrostatic repulsion effects and the stability of NZVI
particles.^{16,20,123,124} However, minimal attention has been given to the impact of stabilizers on
the passivation of NZVI.

In oxygenated solutions, NZVI can undergo four electron transfer steps leading to the 501 formation of reactive species such as HO^{*} and Fe(IV), which can potentially be utilized in the 502 503 oxidative degradation of certain organic contaminants. The addition of iron-chelating agents 504 such as oxalate, nitrilotriacetic acid, ethylenediaminetetraacetic acid, tetrapolyphosphate and 505 polyoxometalate may accelerate the formation of ROS by mediating electron transfer from 506 NZVI to oxygen (e.g., changing the four-electron transfer process to a two-electron transfer process) and by forming soluble complexes with iron at neutral pH values.¹²⁵⁻¹²⁷ Although 507 these studies did not assess the NZVI passivation byproducts in the presence of iron 508 complexing agents and O₂, the acceleration of the NZVI corrosion rate is expected under such 509 510 conditions.

The effects of short chain organic acids (e.g., formic, oxalic and citric acid) on nitrate 511 reduction by Fe(0) has been observed to increase in accordance with the degree to which the 512 ligands adsorb to the surface of iron (oxyhydr)oxides or the stability constants of the aqueous 513 complexes between Fe(III) and the organic ligands.¹¹² The blockage of reactive sites on the 514 515 surface of Fe(0) and its passivation byproducts by the adsorption of ligands through an inner-516 sphere complexation mechanism may be responsible for the observed decrease in reduction 517 rate. Unfortunately, information on whether and/or how the specific adsorption of ligands to 518 the outer layers of corrosion products may alter the ongoing passivation or structural 519 evolution of NZVI is very scarce.

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521 5. DEPASSIVATION STRATEGIES

As discussed in the previous sections, most NZVI studies have focused on the passivation of NZVI after its use in water decontamination reactions. A decrease in NZVI reactivity is considered one of the major issues and a solution is required if NZVI is to be used to effectively degrade contaminants. In this section, a variety of depassivation strategies (including physical, chemical and biological approaches) are introduced to enhance and/or prolong the reactivity of NZVI. Particular attention is given to the use of depassivation strategies before, during and after the decontamination process.

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530 **5.1. Depassivation before decontamination**

A well-known limitation to the broad application of NZVI for contaminant degradation is 531 532 rapid pyrophoric reaction in air and surface passivation in aqueous solutions. This has led to studies examining possible synthesis procedures that can be used to produce reactive NZVI, 533 i.e., NZVI that is atmospherically stable⁸⁶ or stable in aqueous phase.¹²⁸ For example, the 534 shell-modified NZVI particles produced by Kim et al.⁸⁶ after controlled air-oxidation at 535 various flow rates for 1 d did not ignite upon later exposure to atmospheric conditions. 536 Shorter and longer controlled oxidation times resulted in spontaneous combustion or 537 significantly decreased TCE degradation rates, respectively.⁸⁶ The optimal shell-modified 538 539 NZVI particles exhibited a TCE degradation rate that was 78% of that obtained for fresh NZVI (Figure S5). It was also revealed that after aging in water for 1 d, both the fresh and 540 541 shell-modified NZVI exhibited similar TCE degradation rate constants. It was conjectured 542 that surface depassivation of the shell-modified particles was the cause for this observation, 543 however, no mechanisms were examined or postulated.

It has been reported that annealing can induce structural and chemical changes in NZVI, with re-ordering and recrystallization of the inner Fe core, so as to eliminate defects and impurities.¹²⁹ In this case, the thickness of the surface oxide layer decreased from 3–4 nm to 2

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nm by vacuum annealing at 500 °C with an accompanying change of the surface oxide from magnetite to wüstite and the migration of boron and carbon to the particle surface.¹²⁹ This depassivation process was postulated to improve the corrosion resistance and reactive lifespan of NZVI for environmental applications. A subsequent study performed by this group revealed that NZVI reactivity was significantly enhanced (by a factor of 30), with respect to the degradation of TCE and cis-1,2-dichloroethene, when compared to that of fresh NZVI due to efficient electron transfer from Fe(0) to the contaminants.¹³⁰

In recent years there has been an explosion of interest in the sulfidation of NZVI during 554 its synthesis to prevent surface passivation reactions that result from the reduction of water. 555 Using variations of the method first outlined by Kim et al.,¹²⁸ several studies have revealed 556 that sulfidated NZVI is superior to NZVI in decontamination reactions.^{35,128,131} Enhanced 557 decontamination by sulfidated NZVI has been hypothesized to result from the inhibition of 558 corrosion in water.⁶⁵ The formation of nanocrystalline FeS phases on the surfaces of 559 sulfidated NZVI, resulting in a high surface area and good electrical conductivity, has been 560 suggested to be responsible for the high reactivity of these nanoparticles toward contaminants 561 (Figure S6).¹²⁸ From a molecular point of view, dechlorination by reductive elimination may 562 be favored over hydrogenolysis as a result of sulfidation.⁶⁵ In addition, the surface of 563 564 sulfidated NZVI may act as a more efficient conductor of electrons from the inner Fe(0) to the surface layers than iron (oxyhydr)oxides.^{131,132} Two recent review papers have highlighted 565 progress made in this research area.^{65,133} 566

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568 **5.2. Depassivation during/after decontamination**

A significant amount of effort has been expended on the investigation of methods to extend the longevity of NZVI during the removal of environmental contaminants. For example, Bae and Lee⁸ recently demonstrated that an endogenous electron shuttle (i.e.,

riboflavin) can enhance the long-term dechlorination of CCl₄ by NZVI. Results from TEM 572 surface analyses revealed the removal of the passivated layer (Figure S7), presumably due to 573 reductive dissolution by riboflavin, which could result in an increase in the rate of electron 574 transfer. In addition, Wang et al.¹⁴ revealed that the addition of humic acid enhanced the rate 575 576 and extent of 4-chlorobiphenyl degradation by NZVI, given that the humic acid acted as an 577 electron shuttle, thus leading to the rapid transfer of electrons from NZVI to the contaminant. Although the investigators did not investigate the depassivation of NZVI,^{8,14} their 578 experimental results indicated that other electron shuttles such as guinone moieties (e.g., p-579 hydroquinone, lawsone, 9,10-anthraquinone-2,6-disulfonate (AQDS), etc.) may also lead to 580 an increase in the longevity of NZVI. 581

Ultrasound has commonly been used to improve the dispersivity and stability of NZVI during decontamination of wastewater.^{134,135} This process induces cavitation leading to extremely high local temperatures (up to 10000 K) and pressures (up to 5000 bar).¹³⁶ This can cause pitting and cracking at particle surfaces resulting in the removal of the passivated surface (oxyhydr)oxide layer.¹³⁷ The enhanced removal of several elements, including Zn^{2+} , Pb²⁺ and Cu²⁺ has been observed following the application of ultrasound with the increased removal due probably to the removal of the passivated layer.¹³⁶

589 There have been very few studies on the regeneration of passivated NZVI after decontamination¹³⁸ although several studies have been conducted on ZVI.^{85,116,139} Due to the 590 591 different Fe (oxyhydr)oxides that may form on corrosion of NZVI and ZVI and the inherent physical and chemical differences of these particles, it is not appropriate to necessarily 592 assume similar mechanistic processes are operative for ZVI and NZVI. For example, a 593 significant enhancement of TCE reduction was observed when an iron reducing bacteria 594 (Shewanella putrefaciens) was inoculated into a passivated ZVI suspension but not when 595 introduced to NZVI.¹³⁸ It was speculated that the biologically-mediated reductive dissolution 596

597 of the passivating layer re-exposed the ZVI Fe(0) core whereas a minimal amount of Fe(0) 598 was conserved in the case of passivated NZVI. Overall, the lack of studies on NZVI 599 depassivation after decontamination reactions is a result of the small size and highly reactive 600 nature of NZVI, which generally leads to its rapid transformation to Fe(II) and Fe(III) 601 minerals.

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6. ENVIRONMENTAL SIGNIFICANCE AND FUTURE APPLICATIONS

604 NZVI has been one of the most intensively studied nanomaterials in environmental engineering over the past 20 years, particularly in relation to its potential use in remediating 605 606 soil, groundwater and wastewater. Despite the extensive in situ and ex-situ use of NZVI for 607 contaminant removal, rapid surface passivation resulting in loss of NZVI reactivity is considered one of the most severe problems with regard to its effective application. As 608 609 discussed in this review, the nature and extent of NZVI passivation may differ markedly depending on the particular conditions in which NZVI is applied (Figure 6). As such, the need 610 611 exists to investigate the particular passivation mechanism, the factors that influence the rate and extent of passivation and possible depassivation techniques suited to promoting the 612 613 efficacy of NZVI for each proposed application.

Most of the published literature to date has investigated the reactivity of NZVI with very little attention paid to its passivation mechanisms. The reaction between NZVI and contaminants is generally used to assess particle reactivity, however analyses of NZVI transformations have rarely been conducted by investigators. Moreover, passivation effects are more significant when NZVI is applied at a pilot- and/or field-scale due to the presence of many influencing factors (e.g., long reaction time, O_2 levels, presence of inorganic and organic contaminants and biotic influences). A systematic analysis of the effects of

passivation is therefore necessary to provide a comprehensive understanding of NZVI
chemistry during long-term applications (e.g., groundwater and soil remediation).

In this review, the most recent information with respect to the characterization techniques 623 624 for identifying passivation mechanisms have been summarized, in addition to important factors that influence depassivation under different O_2 conditions and thereby extend the 625 626 longevity of NZVI. It becomes clear that there are significant challenges to overcome to 627 further advance NZVI as a remedial agent for environmental pollution. These challenges 628 include: (i) the establishment of advanced time-resolved methodologies to monitor NZVI transformations; (ii) the evaluation of the contribution of passivation byproducts to sequential 629 630 contaminant removal; (iii) the investigation of microbial populations promoted by the 631 passivation byproducts after soil and groundwater remediation and (iv) the development of novel and efficient depassivation strategies during and after NZVI application. 632

633 In engineered systems (such as wastewater treatment), the assessment of NZVI corrosion byproducts and depassivation ways in order to improve and/or maintain contaminant removal 634 performance will be critical for developing a cost-effective NZVI-based technologies and 635 designing large-scale applications. In natural systems (such as soil and groundwater 636 637 environments) where nanoparticles are injected into affected zones, a comprehensive examination of the transformation and mobility of the injected nanoparticles is essential for 638 639 ecological risk assessments. This will improve our ability to develop new simulation tools for 640 accurately predicting the overall performance of NZVI-based technologies as well as the fate 641 of NZVI particles in natural systems.

642

643 ASSOCIATED CONTENT

644 Supporting Information

645 The Supporting Information is available free of charge via on the ACS publications websites

646 at DOI:

647 Figures S1 to S7 (PDF).

648

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Synthesis method	Reaction condition	O ₂ condition	Reaction time	Passivation byproduct	Refs
Fe ³⁺ + NaBH ₄	complete mixing	oxygenated water + oxic condition	3 d	Lepi	38
$FeCl_{2} \bullet 6H_{2}O + NaBH_{2}$	H ₄ no mixing	oxygenated water	5 d	Magn/Magh	77
10013-01120 + NaD114		+ oxic condition	90 d	Lepi	//
	Cl ⁻ (10 mM)	water	2 h	Magn/Magh	80
$Fe_2(SO_4)_3 \cdot 6H_2O + NaBH_4$		water	2 w	Akag/ Magn/Magh	
·		+ oxic condition	16 w	Lepi /Magn/Magh	
$Fe_2(SO_4)_3 \cdot 6H_2O +$	HCO ₃ ⁻ (10 mM)	water	2 h	Magn/Magh	80
NaBH ₄		+ oxic condition	8 w	Goet/ Magn/Magh	
$Fe_2(SO_4)_3 \cdot 6H_2O +$) + NO ₃ ⁻ (10 mM)	water	2 h	Magn/Magh	80
NaBH ₄		+ oxic condition	16 h	Magn/Magh	
	$SO_4^{2-}(10 \text{ mM})$		2 h	Magn/Magh	
$Fe_2(SO_4)_3 \cdot 6H_2O + NaBH_4$		water	2 w	Lepi/Magn/Magh	80
1101114		+ oxic condition	16 w	Goet/Lepi/Magn/Magh	

Table 1. Representative studies reporting passivation byproducts of NZVI under different experimental conditions.

RNIP 10-DP, product from Toda Kogyo Corp., Reduction of Fe(III) oxides by H ₂	$NO_3^-(10 \text{ mM})$ + rotated end-over-end at 22 °C	deaerated water + anoxic condition	6 m	Magn	33
RNIP 10-DP, product from Toda Kogyo Corp. Reduction of Fe(III) oxides by H ₂	$PO_4^{2-}(10 \text{ mM}) + \text{rotated end-over-end}$ at 22 °C	deaerated water + anoxic condition	1 m 6 m	Magn Vivi/Schw	33
NANOFER 25DS, product from NANO IRON (inorganic S based coating), Reduction of Fe(III) oxides by H ₂	Synthetic groundwater (Cl ⁻ (1.95 mN), HCO ₃ ⁻ (1.14 mN), NO ₃ ⁻ (0.15 mN), SO ₄ ²⁻ (0.44 mM))	deaerated water + anoxic condition		Magn	117
NANOFER 25 DS, product from NANO IRON (inorganic S based coating), Reduction of Fe(III) oxides by H ₂	Synthetic groundwater (Cl ⁻ (1.95 mN), HCO ₃ ⁻ (1.14 mN), NO ₃ ⁻ (0.15 mN), SO ₄ ²⁻ (0.44 mM)) + Humic acid (2.5 mg L ⁻¹)	deaerated water + anoxic condition		Iron carbonate hydroxide hydrate	117
FeCl ₃ •6H ₂ O + NaBH ₄	microcystin-LR (5.0 mg L^{-1}) + PO ₄ ²⁻ (0.4 mM) + added H ₂ O ₂ (6.6 mM)	water + oxic condition	4 h	Vivi	34
FeCl ₃ •6H ₂ O + NaBH ₄	microcystin-LR (5.0 mg L ⁻¹) + humic acid (0.4 mM) + added H ₂ O ₂ (6.6 mM)	water + oxic condition	4 h	Lepi	34

$FeCl_3 \bullet 6H_2O + NaBH_4$	microcystin-LR (5.0 mg L^{-1}) + oxalate (0.4 mM) + added H ₂ O ₂ (6.6 mM)	water + oxic condition	4 h	Magn/Lepi	34
NANOFER STAR, product from NANO IRON	synthetic groundwater with an ionic strength of 15 mM	groundwater + anoxic condition	6 d	Amak/Goet/Magn	81
$E_{a}^{3+} + N_{a}DH$	As(V) (2 mg L^{-1}) + synthetic groundwater at pH 5	groundwater	30 d	Magn/Magh/Lepi/Hema	18
Fe + NaBH ₄		+ oxic condition	60 d	Lepi	
Ee^{3+} + NeDU	synthetic groundwater at pH 9	groundwater	30 d	Magn/Magh	18
re + nadn ₄		l IU+ oxic condition	60 d	Magn/Magh	
	U(VI):Fe(0)=1:21, 50 mM NaCl, 5.8	groundwater	3 d	Chuk	
$FeSO_4 \bullet / H_2O + NaBH_4$	mM NaHCO ₃ and 1 mM CaCl ₂ , synthetic groundwater at pH 7	+ anoxic condition	12 m	Chuk/Magn	82
FeCl ₃ •6H ₂ O + NaBH ₄	NO_3^- (100 mg L ⁻¹), HCO ₃ ⁻ (5 g L ⁻¹)	deaerated water + anoxic condition	3 h	Gree(CO ₃)	83
FeCl ₃ •6H ₂ O + NaBH ₄	NO ₃ ⁻ (100 mg L ⁻¹), Humic acid (1 g L ⁻¹ ; DOC = 320 mg L^{-1})	deaerated water + anoxic condition	3 h	Magn	83
FeCl ₃ •6H ₂ O + NaBH ₄	complete mixing at 300 rpm	deaerated water + anoxic condition	3 d	Wust/Goet/Lepi	78

$FeSO_4 \bullet 7H_2O + NaBH_4 +$ carboxymethyl cellulose	$Na_2S_2O_4$ (0.1 M), rolling on a hematology mixer at 15 rpm and pH 7.2	deaerated water + anoxic condition	2 d	Mack	35
FeCl ₂ •4H ₂ O + NaBH ₄	1 μM H ¹⁴ COO ⁻ , pH 6.2–8.3	oxygenated water + oxic condition	1 d	Side	36
RNIP 10-DS, product from Toda America, Inc. Reduction of Fe(III) oxides by H ₂	Dry powder	deaerated water + anoxic condition	<20 d	Mixed-valent Fe(II) and Fe(III) oxide	102
RNIP 10-DP, product from Toda Kogyo Corp. Reduction of Fe(III) oxides by H ₂	Alkaline slurry, pH =10.6	deaerated water + anoxic	2 y	Magn	140

Lepi: Lepidocrocite $\overline{(\gamma - Fe^{III}OOH)}$, Magn: Magnetite $\overline{(Fe^{II_{1}III_{2}}O_{4})}$, Magh: Maghemite $\overline{(\gamma - Fe^{III_{2}}O_{3})}$, Ferr: Ferrihydrite $\overline{(5Fe^{III_{2}}O_{3} \bullet 9H_{2}O)}$, Akag: akageneite (β -Fe^{IIIOOH}), Schw: schwertmannite (Fe^{III}16O₁₆(OH,SO₄)₁₂₋₁₃·10-12H₂O), Wust: wüstite (Fe^{II}O), Amak: amakinite (Fe^{II}(OH)₂), Goet: goethite (α -Fe^{III}OOH), Hema: hematite (α -Fe^{III}₂O₃), Vivi: vivianite (Fe^{II}₃(PO₄)₂·8H₂O), Gree(CO₃): Carbonate green rust: [Fe^{II}₄Fe^{III}₂(OH)₁₂][CO₂·H₂O], Mack: mackinawite (Fe^{II}S), Side: siderite (Fe^{II}CO₃)

40

Figure captions

Figure 1. STEM images and XEDS maps showing NZVI oxidation by oxygen-containing water (a) fresh NZVI, (b) after 30 min and (c) 5 h reaction. (d) The corresponding graph shows the proportions of Fe present as either Fe(0), ferrihydrite or lepidocrocite as determined by linear combination fitting of EXAFS data. Reproduced with permission from ref 36, Copyright 2016, American Chemical Society.

Figure 2. TEM images of NZVI after reaction with water under anoxic conditions at (a) 25 and (b) 80 °C, respectively, and the long-term corrosion results at (c) 25 and (d) 80 °C, showing the proportions of Fe present as either Fe(0), Fe(OH)₂, or magnetite as determined by ⁵⁷Fe Mössbauer Spectroscopy. Reproduced with permission from ref 39, Copyright 2014, American Chemical Society.

Figure 3. A conceptual model on the passivation of NZVI in water showing that the main passivation byproducts include wustite (FeO), goethite (α -FeOOH) and akaganeite (β -FeOOH) under anoxic conditions while crystalline lepidocrocite (γ -FeOOH) with acicular-shaped structures forms under oxic conditions. Reproduced with permission from ref 78, Copyright 2017, Elsevier.

Figure 4. Results of TEM (a and c) and XRD (b and d) of NZVI showing the passivation of NZVI over one month at pH 8 in 5 and 25 mN HCO_3^- suspensions. The XRD diffraction patterns associated with specific mineral phases are noted as α -Fe for Fe(0), M for magnetite, GR for carbonate green rust, and ICH for iron carbonate hydroxide. Reproduced with permission from ref 4, Copyright 2012, American Chemical Society.

Figure 5. Schematic illustration showing the cation-induced depassivation process of aged micro-scale ZVI by Fe(II) dissolution of sheet or shell structures upon metal complexation on the NZVI surface. Cations: M²⁺: Ba²⁺, Sr²⁺, Ca²⁺, Mg²⁺, Mn²⁺, Co²⁺, Fe²⁺, Ni²⁺, Zn²⁺, Pb²⁺, and Cu²⁺. Reproduced with permission from ref 85, Copyright 2014, American Chemical Society.

Figure 6. Schematic illustrations showing the NZVI applications, typical passivation process of NZVI at different conditions, and environmental significance of the passivation.



Figure 1.



Figure 2.



Figure 3.



Figure 4.



Figure 5.



Figure 6.

TOC

