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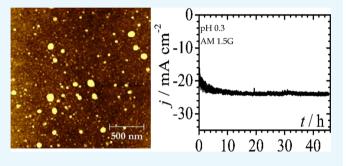
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# Polyoxothiometalate-Derivatized Silicon Photocathodes for Sunlight-Driven Hydrogen Evolution Reaction

Dong Fu,<sup>†</sup> Bruno Fabre,\*<sup>,†</sup> Gabriel Loget,<sup>†</sup> Cristelle Mériadec,<sup>‡</sup> Soraya Ababou-Girard,<sup>‡</sup> Emmanuel Cadot,\*<sup>,§</sup> Nathalie Leclerc-Laronze,<sup>§</sup> Jérôme Marrot,<sup>§</sup> and Quentin de Ponfilly

Supporting Information

ABSTRACT: Silicon photocathodes coated with drop-casted {Mo<sub>3</sub>S<sub>4</sub>}-based polyoxothiometalate assemblies are demonstrated to be effective for sunlight-driven hydrogen evolution reaction (HER) in acid conditions. These photocathodes are catalytically more efficient than that coated with the parent thiomolybdate incorporating an organic ligand, as supported by a higher onset potential and a lower overvoltage at 10 mA cm<sup>-2</sup>. At pH 7.3, the trend is inversed and the beneficial effect of the polyoxometalate for the HER is not observed. Moreover, the polyoxothiometalate-modified photocathode is found to be also more stable under acid conditions and can



be operated at the light-limited catalytic current for more than 40 h. Furthermore, X-ray photoelectron spectroscopy and atomic force microscopy measurements indicate that the cathodic polarization of both photocathodes leads to the release of a large amount of the deposited material into the electrolyte solution concomitantly with the formation of mixed valence species  ${\rm [Mo(IV)_{3-x}Mo(III)_xO_{4-n}S_n]^{(4-x)+}}$  resulting from the replacement of  ${\rm S^{2-}}$  sulfido ligands in the cluster by oxo  ${\rm O^{2-}}$  groups; these combined effects are shown to be beneficial for the photoelectrocatalysis.

## 1. INTRODUCTION

The solar production of hydrogen from the photoelectrochemical reduction of protons has been essentially motivated by the actual growing need for sustainable sources of high-density energies. 1-3 In this thematic area of solar fuels, silicon, with an energy gap of 1.12 eV,4 is a very promising material to be used as a photocathode for the hydrogen evolution reaction (HER) because of its abundance, nontoxicity, and its tunable electronic properties. 5,6 This material fails however to promote efficient multielectron transfers because of slow charge-transfer kinetics.<sup>6,7</sup> Consequently, the deposition of pure metals (the archetypal example is platinum), 6,8-10 metal oxides, 11,12 or sulfides 13-22 onto silicon has been demonstrated to be an effective way to overcome intrinsic limitations of this semiconducting photocathode material for HER.<sup>23</sup>

In that context, silicon derivatized with molybdenum sulfidebased complexes, such as molybdenum disulfide and derivatives  $(MoS_x)^{15,16,18-22}$  or bioinspired cuboidal molecular clusters incorporating the [Mo<sub>3</sub>S<sub>4</sub>]<sup>4+</sup> core, <sup>13,14</sup> has been explored as a cheap and abundant alternative to Pt and the resulting photocathodes were found to exhibit high catalytic efficiency for sunlight-driven HER in acid electrolytes. Interestingly, {Mo<sub>3</sub>S<sub>4</sub>}-based clusters offer rich coordination chemistry able to tune structural, redox, and chemical

reactivity, thus giving the access to a wide range of combinations and properties. <sup>24–27</sup> For instance, over the past decade, the development of synthetic strategies to access hybrid polyoxothiometalates including the {Mo<sub>3</sub>S<sub>4</sub>} core within a polyoxometalate (POM) framework has been engaged. 28,29 The novelty of these systems lies on the combination at the molecular level of an identified active unit, that is, the  $\{Mo_3S_4\}$  core with a POM subunit behaving as electron reservoirs, that is, the POM framework. Because of their high nuclearity, POMs can undergo multielectron processes, making them ideal candidates for engineering complex multistep redox systems (from HER to CO2 reduction).<sup>30,31</sup> Moreover, as POMs are highly functionalizable and generally robust enough to retain their structural integrity during the redox process, we anticipate that such systems coupled to silicon should provide effective and robust photocathodes for sunlight-driven HER. Furthermore, these photocathodes should exhibit enhanced long-term stability compared with that observed for electrodes integrating uncoupled {Mo<sub>3</sub>S<sub>4</sub>} coordination complexes. Indeed, it has

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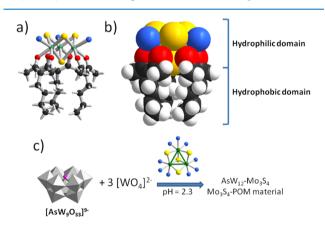
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been reported that the latter ones were susceptible to be degraded in the presence of traces of  $O_2$  in the electrolytic solution.<sup>13</sup>

Herein, silicon photocathodes decorated with {Mo<sub>3</sub>S<sub>4</sub>}-based assemblies incorporating a polyoxotungstate {AsW<sub>12</sub>O<sub>42</sub>} matrix are demonstrated to be effective for sunlight-driven HER under acid pH conditions. Such photocathodes showed a superior catalytic activity and a better robustness compared with photocathodes modified with {Mo<sub>3</sub>S<sub>4</sub>} clusters bearing an organic ligand such as a butylacetylacetonate ligand (noted acac-Bu, hereafter). Moreover, we here investigate the structural and chemical changes at these surfaces that occur during cathodic polarization, which were found to considerably improve the catalytic activity of the photocathodes as well as their stability.

## 2. RESULTS AND DISCUSSION

**2.1.** Characterization of the Molecular Precursors. 2.1.1. Molecular Structure of  $[Mo_3S_4(acac-Bu)_3(H_2O)_3]Cl\cdot H_2O$ . A summary of crystallographic data is given in Table S1. The single-crystal X-ray diffraction analysis of  $Mo_3S_4(acac-Bu)$  revealed the discrete core  $\{Mo_3S_4\}$  coordinated to three butyl-acetylacetonate ligands (acac-Bu) (Figure 1). The



**Figure 1.** Ball and stick (a) and space-filling (b) structural representations of the  $\{Mo_3S_4(H_2O)_3(acac-Bu)_3\}^+$  cationic unit (color code: green sphere = Mo; yellow sphere = S; red sphere = O; blue sphere = aquo ligand; black sphere = C and white sphere = H) showing the hydrophobic part clearly separated from the hydrophilic domain. (c) Scheme of synthesis applied for the preparation of the  $Mo_3S_4$ -POM material resulting from the acidic condensation of tungstate (pH = 2.3) on the trivacant precursor  $[AsW_9O_{33}]^{9-}$  in the presence of the  $[Mo_3S_4(H_2O)_9]^{4+}$  aqua cluster.

coordination of the acac-Bu ligands takes place at the six equivalent equatorial sites of the Mo centers, whereas three aquo ligands were found to be attached to the Mo centers at the axial sites. Geometrical parameters (bond lengths and angle) within the coordination complex were found usual as those previously reported for the analogue  $\{Mo_3S_4\}$ -containing complexes. Nevertheless, the acac-Bu derivative of the triangular cluster  $\{Mo_3S_4\}$  exhibits a striking structural feature corresponding to the relative disposition of the three butyl alkyl chains that run in parallel along the  $C_3$  axis of the cluster (Figure 1b). Such structural characteristics allow distinguishing clearly the hydrophilic polar region, corresponding to the  $\{Mo_3S_4(H_2O)_3\}$  part, whereas the three butyl alkyl chains, arranged as pillars, delimit the hydrophobic domain. Then, amphiphilic behavior could be expected from such a cationic

assembly which could be easily tuned depending on the length of the alkyl chains. Such studies are still ongoing progress.

2.1.2.  $Mo_3S_4$ -POM Materials. The cationic cluster {Mo<sub>3</sub>S<sub>4</sub>}<sup>4+</sup> exhibits a high propensity to be incorporated as a triangular unit within the polyoxometalate framework. This is an elegant way to associate covalently the catalytic  $\{Mo_3S_4\}$ component to an electron reservoir unit. The {Mo<sub>3</sub>S<sub>4</sub>}-POM material was obtained as a mixed rubidium-sodium salt from condensation processes of three equivalents of WO<sub>4</sub><sup>2-</sup> tungstate ions on the trivacant precursor [AsW<sub>9</sub>O<sub>33</sub>]<sup>9-</sup> anion in the presence of  $[Mo_3S_4(H_2O)_9]^{4+}$  at pH 2.3 (Figure 1c). The resulting solid retains the elemental composition Rb<sub>3</sub>NaHAsW<sub>12</sub>Mo<sub>3</sub>S<sub>4</sub>(H<sub>2</sub>O)<sub>3</sub>O<sub>42</sub>·14H<sub>2</sub>O, consistent with the molecular unit  $[AsW_{12}Mo_3S_4(H_2O)_3O_{42}]^{5-}$ . The solid, as well the resulting aqueous solution exhibit a dark-brown color, featuring the coordination of the {Mo<sub>3</sub>S<sub>4</sub>} core to the oxo group of the POM.<sup>29</sup> Thus, the UV-vis spectrum (Figure S1) showed the characteristic weak absorption at 600 nm ( $\varepsilon$  = 520 cm<sup>-1</sup> mol<sup>-1</sup> L) attributed to d-d transitions of the Mo<sup>IV</sup> centers. Furthermore, the IR spectrum of Mo<sub>3</sub>S<sub>4</sub>-POM contains the main absorption related to the polyoxotungstate framework. The vibration band at 950 cm<sup>-1</sup> is attributed to the  $\nu_{as}(W=O)$  mode, whereas those observed at 850 and 798 cm<sup>-1</sup> are assigned to bridging  $\nu_{as}(W-O-W)$  modes.<sup>33,34</sup> Analysis of the IR spectrum of Mo<sub>3</sub>S<sub>4</sub>-POM appears rather consistent with a saturated metal-oxo framework with regard to the wavenumber range of the  $\nu_{as}(W-O-W)$  modes. Vacant POMs give rise to IR vibrations below 800 cm<sup>-1</sup>. Such a result is consistent with the acidic conditions applied for the synthesis procedure (pH = 2.3) which preclude formation of any lacunary species. At last, the Mo<sub>3</sub>S<sub>4</sub>-POM compound has been characterized by <sup>183</sup>W NMR spectroscopy. The <sup>183</sup>W NMR spectrum (Figure S1) shows a complex pattern consisting of more than 20  $^{183}$ W resonances in the -100/-200 ppm chemical shift range differing in their linewidth and relative intensity.<sup>35</sup> This result evidences that the Mo<sub>3</sub>S<sub>4</sub>-POM solid contains a mixture of molecular species which could be isomers or which could differ slightly by their W/ Mo<sub>3</sub>S<sub>4</sub> ratio. Nevertheless, the observed resonances exhibit chemical shifts fully consistent with WVI centers in octahedral oxo environments. The cationic exchange process from alkali to tetra-n-hexylammonium salts alters neither the POM composition nor the IR spectrum (Figure S1).

2.2. Optimization of the Catalyst-Modified Photocathodes for Sunlight-Driven HER. Before investigating the photoelectrocatalytic properties of the best-of-class photocathodes, a preliminary electrochemical study has been carried out to determine the optimal cluster loading leading to maximum catalytic activity. Toward this goal, three cluster loadings [namely,  $7.2 \times 10^{-6}$ ,  $3.6 \times 10^{-5}$ , and  $7.2 \times 10^{-5}$  mmol cm<sup>-2</sup> of Mo<sub>3</sub>S<sub>4</sub>(acac-Bu) or Mo<sub>3</sub>S<sub>4</sub>(AsW<sub>12</sub>)] have been tested for the deposition by drop-casting of the catalyst films onto Si(100)—H. The photoelectrochemical response of the freshly modified surfaces was then examined as a function of the nature and the amount of the deposited catalyst. As expected, no appreciable reduction current was measured in the dark for all electrodes. As shown in Figure S2, the first cyclic voltammetry scans of Mo<sub>3</sub>S<sub>4</sub>(acac-Bu)- and Mo<sub>3</sub>S<sub>4</sub>(AsW<sub>12</sub>)modified Si(100) photocathodes in 1.0 M H<sub>2</sub>SO<sub>4</sub> (pH 0.3) and under simulated sunlight (AM 1.5G, 100 mW cm<sup>-2</sup>) revealed a broad and irreversible cathodic prepeak within the range -0.3/-0.1 V versus reversible hydrogen electrode (RHE), followed by the higher HER photocurrent. It must

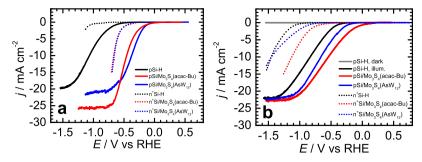


Figure 2. Photoelectrochemical HER at pH 0.3 (a) and 7.3 (b). iR-uncorrected LSV curves at 20 mV s<sup>-1</sup> of optimized catalyst-modified p-type Si(100) photocathodes under simulated sunlight (AM 1.5G, 100 mW cm<sup>-2</sup>) in 1.0 mol L<sup>-1</sup> H<sub>2</sub>SO<sub>4</sub> (a) and 0.1 mol L<sup>-1</sup> KCl + 0.5 mol L<sup>-1</sup> KHCO<sub>3</sub> (b). For comparison, LSV curves of modified n<sup>+</sup>-type Si(100) surfaces in the dark are also shown. The initial cluster loading was  $7.2 \times 10^{-5}$  mmol cm<sup>-2</sup> for Mo<sub>3</sub>S<sub>4</sub>(acac-Bu) and  $7.2 \times 10^{-6}$  mmol cm<sup>-2</sup> for Mo<sub>3</sub>S<sub>4</sub>(AsW<sub>12</sub>).

Table 1. Performances of the Bare and Catalyst-Modified Si(100) Electrodes for Simulated Sunlight-Assisted HER at pH 0.3 and 7.3<sup>a</sup>

cathode	onset potential/ $V_{RHE}^{b}$	$E (E_{corr})/V_{RHE}$ at $-10 \text{ mA cm}^{-2c}$	$\Delta E_{ m ons}/{ m V}^d$	$j_{\rm L}/{\rm mA~cm}^{-2e}$
pH 0.3				
pSi-H	-0.52	-1.07 (-0.92)	0.37	-19.9
pSi/Mo <sub>3</sub> S <sub>4</sub> (acac-Bu)	-0.04	-0.45 (-0.36)	0.27	-25.8
pSi/Mo <sub>3</sub> S <sub>4</sub> (AsW <sub>12</sub> )	-0.02	-0.37 (-0.18)	0.30	-21.5
$n^+Si-H^f$	-0.89	<-1.30		
$n^+$ Si/Mo <sub>3</sub> S <sub>4</sub> (acac-Bu) <sup>f</sup>	-0.31	-0.65 (-0.64)		
$n^+ \text{Si/Mo}_3 \text{S}_4 (\text{AsW}_{12})^f$	-0.32	-0.68 (-0.63)		
		pH 7.3		
pSi-H	-0.32	-0.84 (-0.57)	0.66 (0.45)	-22.0
pSi/Mo <sub>3</sub> S <sub>4</sub> (acac-Bu)	+0.11	-0.54 (-0.22)	0.71 (0.30)	-22.7
$p\mathrm{Si/Mo_3S_4}(\mathrm{AsW_{12}})$	-0.18	-0.66 (-0.44)	0.64 (0.22)	-22.3
$n^+$ Si $-$ H <sup><math>f</math></sup>	-0.98	-1.46 (-1.31)		
n <sup>+</sup> Si/Mo <sub>3</sub> S <sub>4</sub> (acac-Bu) <sup>f</sup>	-0.60	-1.25 (-0.94)		
$n^+Si/Mo_3S_4(AsW_{12})^f$	-0.82	-1.45 (-1.28)		

"Determined from stable photocurrent—potential curves. Determined for a current density of 0.25 mA cm<sup>-2</sup>. Between parentheses, iR-corrected potential values. Difference between onset potentials of illuminated p-type and dark n<sup>+</sup>-type electrodes. Between parentheses (pH 7.3), values obtained when dark n<sup>+</sup>-type electrodes were replaced by glassy carbon ones. Plateau current density corresponding to the light-limited catalytic current. In the dark.

be pointed out that this cathodic prepeak was absent in the following scans and stable sigmoidal photocurrent curves showing the catalytic HER wave were usually obtained after two scans for all tested catalysts and concentrations (Figure 2). The disappearance of such a cathodic prepeak has been previously observed by Jaramillo et al.<sup>36</sup> and Tran et al.<sup>37</sup> for  $[Mo_3S_4]^{4+}$  and  $[Mo_3S_{13}]^{2-}$  cluster-modified surfaces, respectively. It is consistent with either the irreversible conversion of the deposited film into a new material, which is the active form of the catalyst or the cathodic desorption of the cluster into the electrolytic solution, as also proposed by Kristensen et al.<sup>38</sup> In this work, atomic force microscopy (AFM) and X-ray photoelectron spectroscopy (XPS) measurements provide strong experimental evidence that these two events occurred concomitantly upon the electrochemical reduction of the modified photocathodes (vide infra). In the first event, we can thus hypothesize that the reduction of Mo(IV) to Mo(III) is accompanied by the replacement of S<sup>2-</sup> sulfido ligands in the cluster by oxo O<sup>2-</sup> groups, leading to the formation of a mixed O/S mixed valence core  $\{Mo(IV)_{3-x}Mo(III)_xO_{4-n}S_n\}^{(4-x)+}$ , as evidenced by post-electrochemistry XPS analysis (vide infra) and electrochemical measurements of the complexes in solution at a conventional glassy carbon electrode (Figure S4). For the latter, the observed Mo(IV)/Mo(III) reduction potential was very close to that previously reported for

analogous  $[Mo_3S_4]^{4+}$ -based clusters.<sup>39</sup> At pH 7.3, this cathodic prepeak was either much less intense for the  $Mo_3S_4$ (acac-Bu)-modified electrode or no longer observed for the immobilized catalyst  $Mo_3S_4$ (AsW<sub>12</sub>) (Figure S3).

Considering the stabilized catalytic response of the photocathodes, it can be emphasized that surfaces prepared from  $Mo_3S_4(acac-Bu)$  and  $Mo_3S_4(AsW_{12})$  loadings of 7.2 × 10<sup>-5</sup> and  $7.2 \times 10^{-6}$  mmol cm<sup>-2</sup>, respectively, provided the highest photocurrent densities combined to lowest onset potentials (Figure 2). At pH 0.3, the  $Mo_3S_4(AsW_{12})$  cluster was found to be catalytically superior to Mo<sub>3</sub>S<sub>4</sub>(acac-Bu) at low overpotentials (above -0.5 V vs RHE) as the resulting photocathode showed a higher onset potential (-0.02 V vs RHE against -0.04 V) and a lower overvoltage at 10 mA cm<sup>-2</sup> (0.37 V against 0.45 V, corresponding to 0.18 V against 0.36 V when the cell resistance was compensated) (Figure S5). For potentials more negative than -0.5 V versus RHE, the Mo<sub>3</sub>S<sub>4</sub>(acac-Bu)-modified photocathode yielded higher photocurrent densities probably owing to a higher amount of the deposited catalyst. Compared with Si(100)-H illuminated under similar conditions, the decrease in the overpotential for HER measured at 10 mA cm<sup>-2</sup> was approximately 700-740 and 560-620 mV (whether Ohmic drop was corrected or not) for the Mo<sub>3</sub>S<sub>4</sub>(AsW<sub>12</sub>)- and Mo<sub>3</sub>S<sub>4</sub>(acac-Bu)-modified photocathodes, respectively.

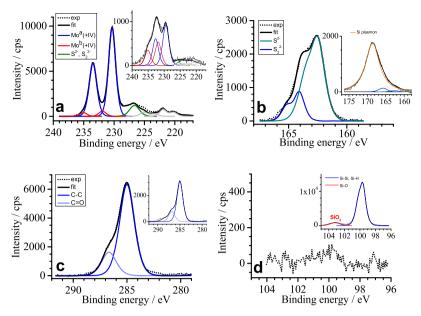


Figure 3. Mo 3d and S 2s (a), S 2p (b), C 1s (c), and Si 2p (d) high-resolution XPS spectra of the freshly prepared  $Mo_3S_4$ (acac-Bu)-modified surface before and after cathodic electrolysis at -0.84 V vs RHE for 2 h under simulated sunlight (insets). The peaks observed below 225 eV (highlighted in gray) are attributed to satellite peaks of the used XPS source. Experimental data and fitting envelopes are represented by black dotted and continuous lines, respectively. The colored lines are fitted curves using Gaussian–Lorentzian mixed peaks corresponding to different components. The initial cluster loading was  $7.2 \times 10^{-5}$  mmol cm<sup>-2</sup>.

Focusing on the potential necessary to reach a current density of 10 mA cm $^{-2}$  as an electrode activity metric, the iR-corrected value determined for the photocathode modified with  $\rm Mo_3S_4(AsW_{12})$  is well in line with  $180{-}200$  mV values reported for highly active  $(\rm Mo_3S_{13})^{2-.40-42}$  MoS $_2$  nanomaterials deposited on planar substrates (e.g., fluorine-doped tin oxide and glassy carbon) exhibited a lower activity for HER as these catalysts achieved 10 mA cm $^{-2}$  at  ${\sim}200{-}250$  mV overpotentials.  $^{40}$  Not surprisingly, MoS $_2$  deposited on high surface area electrodes (graphene oxide, carbon fiber paper, and carbon nanotubes) demonstrated excellent activity characterized by only 110–150 mV overpotential.  $^{40,43-46}$ 

Moreover, from the difference between the onset potentials of the nonphotoactive-modified  $n^+$ -type Si (Figures 2 and S6) and illuminated p-type Si electrodes, it can be deduced that the semiconductor produces a photovoltage of 0.30 V when the deposited catalyst is  $Mo_3S_4(AsW_{12})$  and 0.27 V when the catalyst is  $Mo_3S_4(acac\text{-Bu})$ . Both values are however smaller than that determined for Si(100)–H (i.e., 0.37 V), as a result of enhanced recombination of photogenerated charge carriers.

At pH 7.3, the catalytic performance of the Mo<sub>3</sub>S<sub>4</sub>(AsW<sub>12</sub>)-modified photocathode was found to be lower than that of the Mo<sub>3</sub>S<sub>4</sub>(acac-Bu)-modified photocathode irrespective of the examined potential range, as supported by a lower onset potential (-0.18 V against +0.11 V) and a higher overpotential at 10 mA cm<sup>-2</sup> (0.66 V against 0.54 V, corresponding to 0.44 V against 0.22 V when the cell resistance was compensated) (Table 1). Moreover, the difference between the onset potentials of the nonphotoactive-modified n<sup>+</sup>-type Si and illuminated p-type Si electrodes yielded clearly abnormally high photovoltage values, exceeding 0.60 V for both types of photocathodes. More reasonable values ranging from 0.22 to 0.45 V were determined when n<sup>+</sup>-type Si electrodes were replaced by more conventional glassy carbon ones (Figure S7). Such results could be partly explained by differences in the

catalyst-support interactions and/or surface energetics<sup>47</sup> of  $n^+$ - and p-doped silicon at neutral pH.

**2.3.** Electrochemistry-Induced Structural Changes. The chemical composition and the morphology of the asdeposited and electrochemically reduced cluster films were examined by means of XPS and AFM. First, XPS measurements have been performed on the two types of optimized photocathodes to gain insights about the composition and molecular structure of the as-deposited films and about the nature of the catalytically active species. With the main carbon C 1s peak set at 285.0 eV, the characteristic peaks related to the cluster core are observed.

The freshly prepared Mo<sub>3</sub>S<sub>4</sub>(acac-Bu)-modified surface shows the main Mo  $3d_{5/2}$  and Mo  $3d_{3/2}$  lines at 230.3 and 233.4 eV, respectively, which are consistent with molybdenum in the formal +4 state (Figures 3 and S8). 48 Additionally, two minor components of Mo 3d are observed at 231.9 and 235.1 eV, which can be ascribed to Mo(IV+) atoms in a different local environment, as previously reported by Jaramillo et al.<sup>36</sup> The presence of sulfur is evidenced by the S 2s signal at 226.6 eV. The S 2p region shows a main peak at 162.6 eV corresponding to unresolved contributions of  $2p_{1/2}$  and  $2p_{3/2}$ doublet of S<sup>2-</sup> ligands. A second minor peak at 164.1 eV is also observed, which suggests the concomitant presence of  $S_2^2$ groups. Such observations are in perfect line with previous reports of molybdenum sulfide materials, which incorporate sulfur atoms in different oxidation states. 49,50 The experimental ratio between the areas under the Mo and S peaks is estimated at 0.75, as theoretically expected. Moreover, the C 1s spectrum shows a main component at 285.0 eV and a shoulder at 286.8 eV, which are attributed to C-C and C=O bonds of the (acac-Bu) ligand, respectively. The corresponding C-C/C-O peak area ratio of 3.8 was also fully consistent with the expected atomic composition of the acac-Bu ligand (i.e. 7/2).

Now, for the freshly prepared  $Mo_3S_4(AsW_{12})$ -modified surface, the two Mo  $3d_{5/2}$  peaks attributed to locally different

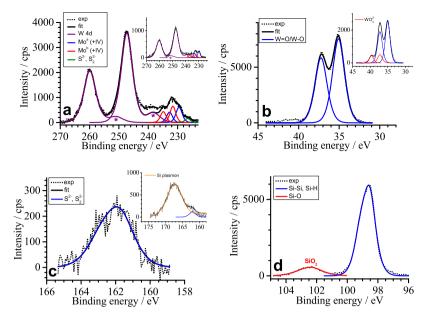


Figure 4. W 4d, Mo 3d, and S 2s (a), W 4f (b), and S 2p (c) high-resolution XPS spectra of the freshly prepared  $Mo_3S_4(AsW_{12})$ -modified surface before and after cathodic electrolysis at -0.84 V vs RHE for 2 h under simulated sunlight (insets). Post-electrolysis Si 2p spectrum (d). The peaks observed at 238.3 and 251.1 eV (indicated by asterisks) are attributed to satellite peaks of the used XPS source. Experimental data and fitting envelopes are represented by black dotted and continuous lines, respectively. The colored lines are fitted curves using Gaussian—Lorentzian mixed peaks corresponding to different components. The initial cluster loading was  $7.2 \times 10^{-6}$  mmol cm<sup>-2</sup>.

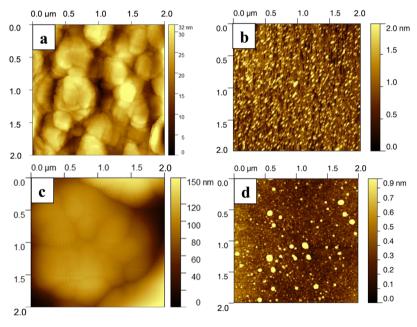


Figure 5. AFM images of  $Mo_3S_4$ (acac-Bu)- (a,b) and  $Mo_3S_4$ (AsW<sub>12</sub>)- (c,d) modified Si(100) surfaces before (a,c) and after cathodic electrolysis at  $-0.84~\rm V$  vs RHE for 2 h under simulated sunlight (b,d). The initial cluster loading of optimized surfaces was  $7.2\times10^{-5}~\rm mmol~cm^{-2}$  for  $Mo_3S_4$ (acac-Bu) and  $7.2\times10^{-6}~\rm mmol~cm^{-2}$  for  $Mo_3S_4$ (AsW<sub>12</sub>). Scan size:  $2\times2~\mu m^2$ .

Mo(4+) atoms are observed at 229.5 and 231.6 eV (Figures 4 and S9). Furthermore, the presence of S<sup>2-</sup> and S<sub>2</sub><sup>2-</sup> groups is also evidenced by the S 2s and S 2p signals at 226.6 and 161.9 eV, respectively. The ratio between the peak areas of the total Mo and S signals is 0.74, as theoretically expected. The W 4f and W 4d lines observed at 35.1 (W 4f<sub>7/2</sub>), 37.2 (W 4f<sub>5/2</sub>), 247.3 (W 4d<sub>5/2</sub>), and 259.9 (W 4d<sub>3/2</sub>) eV are consistent with W-O/W=O species involving tungsten in the formal +6 state. S1-53 The experimental ratio Mo  $3d_{5/2}/W$   $4d_{3/2}$  is estimated at 0.23, in close agreement with the theoretical ratio (i.e., 0.25).

After cathodic electrolysis for 2 h with the deposited clusters, both the intensity and the position of the characteristic XPS signals were strongly modified. For both photocathodes, a large amount of the deposited material has been released into the electrolyte solution during the electrochemical test (about 90 and 70% for Mo<sub>3</sub>S<sub>4</sub>(acac-Bu)- and Mo<sub>3</sub>S<sub>4</sub>(AsW<sub>12</sub>)-modified surfaces, respectively), as confirmed by the dramatic decrease in the intensity of Mo 3d, W 4d, W 4f, and S 2p XPS signals concomitantly with the appearance of both the Si plasmon loss peak and Si 2p signal (insets in Figures 3 and 4). Such observations are in line with those reported for other  $\{Mo_3S_4\}$ -

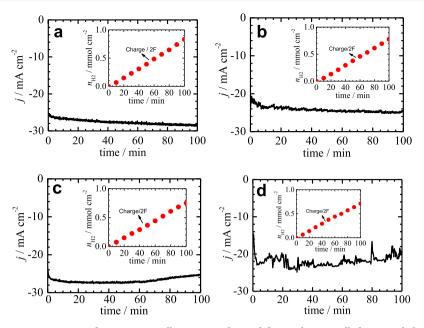


Figure 6. Photocurrent density—time curves under continuous illumination obtained during the controlled potential electrolysis at -0.84 V vs RHE in a pH 0.3 solution (a,b) or -1.04 V vs RHE in a pH 7.3 solution (c,d) using optimized  $Mo_3S_4(acac-Bu)$  (a,c) or  $Mo_3S_4(AsW_{12})$  (b,d)-modified photocathodes. (Insets) Amount of  $H_2$  evolved at the photocathode as a function of the electrolysis time. The mole number of produced gas was calculated from the gas volume trapped in a Hoffman cell. The dotted black lines correspond to the theoretical fit assuming a 100% Faradaic efficiency. The current fluctuations are caused by the removal of  $H_2$  bubbles from the photocathode surface. The surface areas of the photoelectrodes are 0.30 (a), 0.50 (b), 0.49 (c), and 0.51 (d) cm<sup>2</sup>.

based cluster-modified surfaces. 13,38 As a matter of fact, the H<sub>2</sub> bubbling at the surface of the photocathode provokes mechanical desorption of a large upper part of the immobilized layer which does not participate in the catalytic process. Moreover, the main Mo  $3d_{5/2}$  and Mo  $3d_{3/2}$  lines of the electrochemically reduced Mo<sub>3</sub>S<sub>4</sub>(acac-Bu)-modified surface were shifted to 229.4 and 232.6 eV, respectively, which would be consistent with partial substitution of S<sup>2-</sup> sulfido ligands by O<sup>2-</sup> oxo groups in the metallic cluster. Such an assumption is supported by the XPS spectra of  $[Mo_3O_{4-n}S_n]^{4+54}$  or refers to the presence of Mo(III+) in the cluster structure. 55 The Mo 3d component having the highest energy (assigned to Mo(IV+) atoms in a different local environment) was not affected by the electrolytic treatment, both in intensity and position. This leads to the hypothesis that this component could be the signature of the surface-confined, stable, and catalytically active clusters. For the cathodically reduced Mo<sub>3</sub>S<sub>4</sub>(AsW<sub>12</sub>)-modified surface, the Mo 3d lines of the two components were only 0.3 eV shifted toward more positive binding energies. Such a result would be rather consistent with a weaker lability of S<sup>2-</sup> ligands in the cluster incorporating the POM. Additionally, the W 4f spectrum shows an additional component at higher binding energies (37.2 and 39.6 eV for the  $4f_{7/2}$  and  $4f_{5/2}$  levels, respectively) that can be assigned to the formation of some tungstate species.<sup>56</sup> Finally, the Si 2p spectrum shows the presence of a substantial amount of silicon oxides at about 102-103 eV for both cathodically reduced photocathodes. However, it could not be ruled out that the oxidation of the silicon surface was either caused by the cathodic electrolysis or already present after the catalyst deposition because the Si 2p signal for the freshly prepared photocathodes was totally attenuated because of the thick catalyst layer.

From XPS data and using the treatment reported by Cicero et al.,  $^{57}$  if one considers the Si(100) surface atom density of 6.8  $\times$  10<sup>14</sup> atoms cm<sup>-2</sup>,  $^{58}$  the surface coverages of the catalyst

remaining after cathodic electrolysis can be estimated at  $(8.0 \pm 0.5) \times 10^{13}$  and  $(2.0 \pm 0.6) \times 10^{13}$  cm<sup>-2</sup> for the Mo<sub>3</sub>S<sub>4</sub>(acac-Bu)- and Mo<sub>3</sub>S<sub>4</sub>(AsW<sub>12</sub>)-modified photocathodes, respectively (see the Supporting Information).

Electrochemistry-induced changes derived from XPS were also corroborated by the AFM analysis of the optimized photocathodes. After the catalyst deposition, both surfaces showed a rough and compact granular structure characterized by the presence of a few hundreds of nanometer large agglomerates (Figure 5). The root-mean-square (rms) roughness was measured at ca.  $6.0 \pm 0.5$  and  $26.0 \pm 2$  nm for the Mo<sub>3</sub>S<sub>4</sub>(acac-Bu)- and Mo<sub>3</sub>S<sub>4</sub>(AsW<sub>12</sub>)-modified surfaces, respectively. After cathodic electrolysis, a much less dense array was observed with the presence of a few tens of nanometer large catalyst aggregates and the surface of underlying silicon became now visible. The rms roughness values were therefore considerably decreased and lower than 1.0 nm for both surfaces. The processing of AFM images yielded cluster surface density values of  $(4.7 \pm 0.8) \times 10^{13}$  and  $(6.4 \pm 0.8) \times 10^{12}$ cm<sup>-2</sup> for the Mo<sub>3</sub>S<sub>4</sub>(acac-Bu)- and Mo<sub>3</sub>S<sub>4</sub>(AsW<sub>12</sub>)-modified surfaces, respectively, considering the theoretical projected specific area of one cluster (Figures S10 and S11). Different reasons can be involved to explain such discrepancies between the cluster densities extracted from XPS and AFM data, especially for the Mo<sub>3</sub>S<sub>4</sub>(AsW<sub>12</sub>)-modified surface. AFM is believed to be a less reliable technique than XPS to estimate such a parameter essentially due to large uncertainties arising from (i) the calculation method based on the microscale analysis of the surface, (ii) the used processing method that is based on the detection of a size threshold to determine the surface area covered by the cluster, and (iii) the differences between the experimental and theoretical specific areas occupied by the cluster.

2.4. Faradaic Efficiency and Turnover Frequency of the Photocathodes for Solar-Driven HER. The cluster-

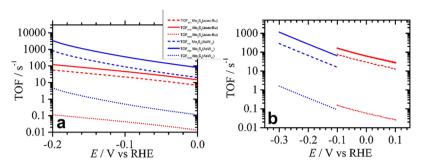


Figure 7. TOF of the  $Mo_3S_4$ (acac-Bu) (red lines) and  $Mo_3S_4$ (AsW<sub>12</sub>) (blue lines) clusters deposited on p-type Si(100) surfaces for simulated sunlight-driven HER at pH 0.3 (a) and 7.3 (b). The minimum (TOF<sub>min</sub>, dashed lines) and maximum (TOF<sub>max</sub>, solid lines) TOF numbers were determined from upper and lower limits of the post-electrolysis cluster densities, respectively, estimated from XPS and AFM data. TOF<sub>load</sub> (dotted lines) was calculated from the initial loading of the deposited cluster.

supported photoelectrodes exhibiting the best electrocatalytic efficiencies [namely, those prepared from a cluster loading of  $7.2 \times 10^{-5} \text{ mmol cm}^{-2} \text{ for Mo}_3 S_4 (\text{acac-Bu}) \text{ or } 7.2 \times 10^{-6}$ mmol cm $^{-2}$  for Mo $_3S_4(AsW_{12})$ ] were tested for more than 1 h in both electrolytic media under simulated sunlight. As shown in Figure 6, the photocurrent densities measured during the controlled potential electrolysis tests were in quite good agreement with those derived from linear sweep voltammetry (LSV) curves. Moreover, a careful examination of all chronoamperometric curves shows a short initial period (about 5 min) for which the cathodic current increases before stabilizing. This beneficial effect results probably from the loss of a large amount of deposited clusters, as supported by the post-electrolysis XPS and AFM data (vide supra), leading to both a better light collection and higher conductivity at the solid-liquid interface.

At pH 0.3, the photocurrent densities measured for both photoelectrodes remained relatively stable during the electrolysis time (Figure 6). The amount of evolved  $H_2$  was determined from the gas volume by measuring the amount of the electrolytic solution displaced in a sealed and graduated upside-down burette (Hoffman cell).<sup>59</sup>  $H_2$  was quantitatively produced (Faradaic efficiency of 98  $\pm$  2%) with a rate of 0.48  $\pm$  0.02 mmol cm<sup>-2</sup> h<sup>-1</sup>, irrespective of the nature of the tested cluster. At pH 7.3, both photocathodes were less stable but the production rate of  $H_2$  was almost unchanged. Noticeably, the Faradaic efficiency for the photocathode incorporating  $Mo_3S_4(AsW_{12})$  was higher than that with  $Mo_3S_4(acac-Bu)$ , 99  $\pm$  1% against 91  $\pm$  1%.

To quantify the catalytic efficiency of a system, the turnover frequency (TOF) as a function of the electrode potential has been determined. 40,60 For a HER catalyst, the TOF corresponds to the ratio of the number of H<sub>2</sub> moles produced per second to the number of moles of catalytically active sites (see the Supporting Information for more details). It is therefore essential that the contribution of the electrode surface in the catalytic process is eliminated. Consequently, in this work, this quantity was determined from iR-corrected LSV curves of the optimized Mo<sub>3</sub>S<sub>4</sub>(acac-Bu)- and Mo<sub>3</sub>S<sub>4</sub>(AsW<sub>12</sub>)modified photocathodes (Figure S5) within a potential window corresponding to the foot of the catalytic wave, 61 that is, -0.2 to 0.0 V versus RHE (for both clusters at pH 0.3), -0.1 to 0.1 V versus RHE [for the Mo<sub>3</sub>S<sub>4</sub>(acac-Bu) cluster at pH [7.3], or [-0.3] to [-0.1] V versus RHE [for the Mo<sub>3</sub>S<sub>4</sub>(AsW<sub>12</sub>) cluster at pH 7.3]. It is noteworthy that limitations by the mass transport can be considered as negligible owing to the high concentration of the electrolyte.

Moreover, another critical issue is the determination of the exact number of the active sites present at the electrode surface because it is quite obvious that an underestimation of this quantity will yield an overestimation of the TOF. In this work and according to previous reports on thiomolybdate-based HER electrocatalysts, 13,36 the number of active sites should correspond to the cluster density remaining after the cathodically induced desorption step and estimated from two different methods, namely XPS and AFM. Thereby, this is supported by the stability of both the photocurrents measured at pH 0.3 during preparative-scale electrolysis for a few tens of hours and by the LSV response after electrolysis, as shown in the next section. Nevertheless, for comparison, TOF values calculated using the initial loading of the deposited cluster are also considered. At pH 0.3, the variation of TOF with the electrode potential confirms the higher catalytic activity of the cluster Mo<sub>3</sub>S<sub>4</sub>(AsW<sub>12</sub>) compared to Mo<sub>3</sub>S<sub>4</sub>(acac-Bu) (Figure 7). From the post-electrochemistry cluster density, TOF values in the range 20-80 and 6-14 s<sup>-1</sup> were estimated at 0 V versus RHE for the Mo<sub>3</sub>S<sub>4</sub>(AsW<sub>12</sub>) and Mo<sub>3</sub>S<sub>4</sub>(acac-Bu) clusters, respectively. Respective values of ca. 850-3300 and 50-120  $s^{-1}$  were reached at -0.2 V versus RHE. At pH 7.3, the trend was inversed with Mo<sub>3</sub>S<sub>4</sub>(acac-Bu) being more active than Mo<sub>3</sub>S<sub>4</sub>(AsW<sub>12</sub>) (Figure 7b). The TOF values fall in the 60- $250 \text{ s}^{-1}$  range for Mo<sub>3</sub>S<sub>4</sub>(AsW<sub>12</sub>) at an overpotential of 0.2 V, whereas Mo<sub>3</sub>S<sub>4</sub>(acac-Bu) yielded 30-60 s<sup>-1</sup> TOF values at 0 V versus RHE.

Such exceptional values have never been reported for thiomolybdate-based HER electrocatalysts 41,62 so far and exhibit ca. 2-3 orders of magnitude higher than those reported for the most active molybdenum sulfides, such as MoS<sub>2</sub> nanoparticles supported on Au(111).<sup>41,63</sup> Much smaller TOF values [for instance, ca. 6 and  $0.1 \text{ s}^{-1}$  for Mo<sub>3</sub>S<sub>4</sub>(AsW<sub>12</sub>) and Mo<sub>3</sub>S<sub>4</sub>(acac-Bu), respectively, at −0.2 V versus RHE and pH 0.3] competing with that of the best molybdenum sulfidebased HER catalysts<sup>41</sup> are calculated if the number of active cluster sites is assumed to be equal to the initial loading of the deposited cluster (Figure 7). As a matter of fact, a mean TOF between these two extreme limits should be rather representative of the real catalytic activity of the immobilized cocatalysts. Importantly, a special care must be taken to the comparison of our TOF values estimated at a certain overpotential with those determined for other molybdenum sulfide- or thiomolybdate-based HER electrocatalysts because all reported TOF values up to date concern catalysts supported on surfaces other than semiconductors. Additionally, a few studies devoted to other molecular HER electrocatalysts

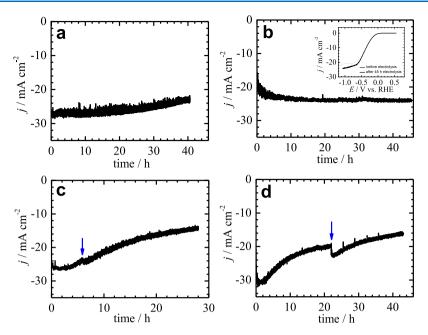


Figure 8. Long-term stability of the optimized  $Mo_3S_4(acac-Bu)$  (a,c) and  $Mo_3S_4(AsW_{12})$  (b,d)-modified photocathodes for simulated sunlight-driven HER during the controlled potential electrolysis at -0.84 V vs RHE in a pH 0.3 solution (a,b) or -1.04 V vs RHE in a pH 7.3 solution (c,d). (Inset) iR-uncorrected LSV curves at 20 mV s<sup>-1</sup> (pH 0.3) of the  $Mo_3S_4(AsW_{12})$ -modified photocathode before and after electrolysis. The blue arrows indicate the removal of the  $H_2$  bubbles from the surface.

attached to p-type semiconductors have reported an order of magnitude lower TOFs. For instance, metalloporphyrin covalently bound to p-type GaP yielded a TOF of ca. 4 s<sup>-1</sup> at 0 V versus RHE and pH 7 under identical illumination conditions.<sup>64</sup> In another example, a cobalt molecular complex bound to a TiO<sub>2</sub>-protected GaInP<sub>2</sub> photocathode produced H<sub>2</sub> with a TOF of 3.4 s<sup>-1</sup> at 0 V versus RHE but under basic conditions (pH 13).65 Besides the possible underestimation of the density of catalytically active sites, the high TOF values reported in this work may also be due to a significant contribution of the silicon surface in the catalytic process even though special care has been taken to measure accurately the TOF of the alone immobilized catalyst (restricted potential range in which the current densities are expected to be controlled by the HER kinetics and the use of high concentration of the electrolyte).

2.5. Long-Term Stability of the Photocathodes. The stability criterion is another crucial requirement for modified photocathodes, mainly driven by the robustness of the interface. To assess the long-term stability of our photocathodes, controlled potential electrolyses were carried out in both aqueous solutions over testing periods exceeding ca. 30 h. At pH 0.3, the Mo<sub>3</sub>S<sub>4</sub>(AsW<sub>12</sub>)-modified photocathode was found to be the most robust because no noticeable decrease in the photocurrent was observed after 40 h. In contrast, a 18% decrease in the cathodic photocurrent was measured for the Mo<sub>3</sub>S<sub>4</sub>(acac-Bu)-modified photocathode (Figure 8). Such a trend features the high stability of the Mo<sub>3</sub>S<sub>4</sub>(AsW<sub>12</sub>) POM material functioning under acidic conditions. The similarity between the LSV curves before and after electrolysis supports the excellent stability of the Mo<sub>3</sub>S<sub>4</sub>(AsW<sub>12</sub>)-modified photocathode. At pH 7.3, in line with short-term stability tests detailed in the previous section, both photocathodes were found to be much less stable during the potentiostatic electrolysis with a comparable decrease in the photocurrent after 30 h, namely, 38% against 45%. We believe that the lower stability and lower activity observed at pH 7.3 for both modified photocathodes can be ascribed to some decomposition reactions of the immobilized catalysts and/or a different Si chemical interface. In the case of the  $Mo_3S_4$ (acac-Bu) catalyst, the first process may originate from the substitution of the organic ligand by some anions from the electrolyte, whereas a hydrolytic decomposition leading to the transformation (such as partial decondensation) of the  $\{Mo_3S_4-POM\}$  hybrid material is possible in the case of the second catalyst. The second process associated to the electrode may lead to the gradual degradation of the Si/catalyst interface or could enhance catalyst desorption.

## 3. CONCLUSIONS

In this work, we have demonstrated that a p-type silicon photocathode coated with a polyoxothiometalate electrocatalyst was catalytically more efficient than a photocathode coated with the parent thiomolybdate incorporating an organic ligand for the simulated sunlight-driven HER under acid pH conditions. These findings were supported by a higher onset potential and a lower overvoltage at 10 mA cm<sup>-2</sup>, in spite of a ca. 4–7 times smaller catalyst coverage. Under such conditions, the polyoxothiometalate-modified photocathode produced H<sub>2</sub> quantitatively with a promising stability along 40 h controlled potential electrolysis tests. In contrast, the photocurrents measured at a thiomolybdate-modified surface were gradually ca. 20% decreased under the same conditions. At neutral pH, the situation was inversed and the beneficial effect of the POM for the HER was not observed.

It is anticipated that the catalytic performance of our electrodes could be improved by using Si-structured surfaces incorporating an additional n<sup>+</sup> Si layer on the p-type Si surface of the photocathode structure, as recently proposed for CoP-coated HER photocathodes.<sup>68</sup> Moreover, very rich chemistries of POMs and chalcogenide clusters are also promising directions to improve further HER activity of this new class

of Mo<sub>3</sub>S<sub>4</sub>–POM electrocatalysts.<sup>69,70</sup> Finally, because of their high metal nuclearity, POMs represent relevant molecular systems for multielectron transfer engineering and then ideal candidates to tackle the challenging electrocatalytic process of great societal interest, namely, the CO<sub>2</sub> reduction.

## 4. EXPERIMENTAL SECTION

**4.1. Reagents.** Acetone (MOS electronic grade, Erbatron from Carlo Erba), anhydrous ethanol (RSE electronic grade, Erbatron from Carlo Erba), and 3-butylpentane-2,4-dione (Sigma-Aldrich) were used without further purification. The cationic cluster {Mo<sub>3</sub>S<sub>4</sub>}<sup>4+</sup> was prepared through a previously reported procedure and available in the solid state as (NH<sub>4</sub>)<sub>2</sub>Mo<sub>3</sub>S<sub>4</sub>(H<sub>2</sub>O)<sub>9</sub>Cl<sub>6</sub>·3H<sub>2</sub>O. The chemicals used for the cleaning and etching of silicon wafer pieces (30% H<sub>2</sub>O<sub>2</sub>, 96–97% H<sub>2</sub>SO<sub>4</sub> and 50% HF solutions) were of VLSI (H<sub>2</sub>O<sub>2</sub>, from Sigma-Aldrich) and MOS (H<sub>2</sub>SO<sub>4</sub> from O-BASF and HF from Sigma-Aldrich) semiconductor grade.

**4.2. Syntheses of the Compounds.** 4.2.1. [Mo<sub>3</sub>S<sub>4</sub>(acac-Bu)<sub>3</sub>(H<sub>2</sub>O)<sub>3</sub>]Cl·H<sub>2</sub>O. (NH<sub>4</sub>)<sub>2</sub>Mo<sub>3</sub>S<sub>4</sub>(H<sub>2</sub>O)<sub>9</sub>Cl<sub>6</sub>·3H<sub>2</sub>O (1.0 g; 1.13 mmol) was dissolved in methanol (30 mL). 3-Butylpentane-2,4-dione (noted acac-Bu) (0.695 g; 4.45 mmol) and 2.2 mL of 1 mol L<sup>-1</sup> KOH aqueous solution were mixed in 5 mL of methanol and then slowly added to the greenish {Mo<sub>3</sub>S<sub>4</sub>}-containing solution. The resulting darkgreen solution was allowed to stand for crystallization. After 2 days, well-shaped parallelepipedic crystals suitable for X-ray diffraction analysis were collected. Yield: 0.66 g (60% vs Mo). <sup>1</sup>H NMR (300 MHz, MeOD): δ (ppm) 8.89 (t, 3H), 1.29 (2 × 2H), 2.11 ppm (6H), 2.29 (2H). Anal. Calcd for Mo<sub>3</sub>S<sub>4</sub>C<sub>29</sub>H<sub>55</sub>O<sub>10</sub>Cl: Mo, 0.28; S, 0.12; C, 0.33; H, 0.054; Cl, 0.03. Found: Mo, 0.27; S, 0.11; C, 0.36; H, 0.058; Cl, 0.03.

4.2.2.  $Rb_3NaHAsW_{12}Mo_3S_4(H_2O)_3O_{42}\cdot 14H_2O$ . Sodium tungstate dihydrate Na<sub>2</sub>WO<sub>4</sub>·2H<sub>2</sub>O (1.11 g; 3.36 mmol) and Na<sub>9</sub>AsW<sub>9</sub>O<sub>33</sub>·12H<sub>2</sub>O (3.0 g; 1.12 mmol) were suspended in 30 mL of water and acidified by 3.5 mL of aqueous solution of 2 mol  $L^{-1}$  hydrochloric acid until pH was down to about pH = 5. The resulting solution was then heated at 50 °C while 10 mL of aqueous solution containing (NH<sub>4</sub>)<sub>2</sub>Mo<sub>3</sub>S<sub>4</sub>(H<sub>2</sub>O)<sub>9</sub>Cl<sub>6</sub>· 3H<sub>2</sub>O (1.0 g; 1.13 mmol) was slowly added drop by drop under gentle stirring. A cloudy brownish precipitate was formed and disappeared slowly after 30 min at 50 °C. The pH of the solution was 2.3. After the solution cooled down to room temperature, rubidium chloride (1.5 g; 12 mmol) was poured to the solution provoking the formation of a darkbrown solid. After 30 min under slow stirring, the solid was then collected through filtration using a glass frit filter, washed with ethanol, and dried using diethyl ether. Analysis of the resulting solid (2.8 g) was consistent with Rb<sub>3</sub>NaHAsW<sub>12</sub>Mo<sub>3</sub>S<sub>4</sub>(H<sub>2</sub>O)<sub>3</sub>O<sub>42</sub>·14H<sub>2</sub>O, leading to a yield of about 65%. Anal. Calcd for Rb<sub>3</sub>NaH<sub>35</sub>AsW<sub>12</sub>Mo<sub>3</sub>S<sub>4</sub>O<sub>59</sub>: Rb, 6.5; Na, 0.6; As, 1.9; W, 55.7; Mo, 7.3; S, 3.2. Found: Rb, 6.6; Na, 0.55; As, 1.8; W, 55.7; Mo, 7.2; S, 3.3. Thermal gravimetric analysis (TGA): weight loss of 8% between RT and 200 °C corresponding to 17.5 water molecules distributed as 14 crystallization water molecules and 3 aquo ligands attached to the  $\{Mo_3S_4\}$  core and 1/2 constitutional water. IR  $(cm^{-1})$ : 950(s), 847(m), 780(vs), 656(w), 445(w).

4.2.3.  $[(n-C_6H_{13})_4N]_5AsW_{12}Mo_3S_4(H_2O)_3O_{42}$ . The cationic exchange has been carried out through liquid—liquid extraction of the POM as an alkali salt from the aqueous solution to the toluene organic phase containing tetra-n-hexylammonium bromide. The 50 mL aqueous solution contained 2.55 mmol

L<sup>-1</sup> in Rb<sub>3</sub>NaHAsW<sub>12</sub>Mo<sub>3</sub>S<sub>4</sub>(H<sub>2</sub>O)<sub>3</sub>O<sub>42</sub>·14H<sub>2</sub>O, whereas the organic phase corresponded to tetrahexylammonium bromide (0.40 g; 0.92 mmol) dissolved in 40 mL of toluene. Both solutions were contacted vigorously for 30 min and then decanted within a separatory funnel. The brownish organic phase was collected. The solvent was removed through a rotary evaporator until obtaining a dark-brown oily residue which was thoroughly washed with ethanol and diethylether. Yield: 340 mg (46%). Anal. Calcd for C, 27.7; H, 5.0; N, 0.9. Found: C, 28.3; H, 5.1; N, 1.1; EDX atomic ratios. Calcd (found): W/Mo = 4(3.9); W/S = 3(2.8). TGA: weight loss of 35% between RT and 500 °C in agreement with the combustion of five tetra-*n*-hexylammonium units. IR (cm<sup>-1</sup>): 957(s); 850(m); 797(vs); 729(sh).

**4.3.** Analytical Methods for Characterization of the Precursors. The NMR spectra were measured in  $D_2O$  at 27 °C.  $^1H$  and  $^{183}W$  NMR spectra were recorded on a Bruker AVANCE 500 spectrometer at 500.1 and 20.84 MHz, respectively. According to conventions for standards, chemical shifts are referenced with respect to TMS and 1 M  $Na_2WO_4$  aqueous solution for  $^1H$  and  $^{183}W$ , respectively.

Fourier transform infrared (FT-IR) spectra were recorded on a 6700 FT-IR Nicolet spectrophotometer, using the diamond attenuated total reflection (ATR) technique. The spectra were recorded on nondiluted compounds in the range 400–4000 cm<sup>-1</sup>. ATR correction was applied.

UV—vis spectra of powdered compounds have been collected by using a PerkinElmer LAMBDA 750 spectrophotometer equipped with a powder sample holder set. The UV—vis spectra of solutions were recorded on PerkinElmer LAMBDA-750 using a calibrated 0.1 cm quartz-cell.

Energy-dispersive X-ray spectra were obtained using a scanning electron microscope enhanced by a field emission gun equipment (JSM 7001-F, JEOL). The measurements were acquired with a SDD X-Max 50 mm<sup>2</sup> detector and the Aztec (Oxford) system working at 15 kV and 10 mm distance. The quantification is realized with the standard library provided by the constructor using L $\alpha$  lines.

The number of water molecules and tetra-*n*-hexylammonium cations were determined by TGA with a Mettler Toledo TGA/differential scanning calorimetry 1, STAR<sup>e</sup> System apparatus or with a NETZSCH TG 209 F1 device under air flow (50 mL min<sup>-1</sup>) at a heating rate of 5 °C min<sup>-1</sup> up to 800 °C.

4.4. Single-Crystal X-ray Diffraction Study. Intensity data collections were carried out at T = 200(2) K with a Bruker X8 APEX2 charge-coupled device, using the MoK $\alpha$ wavelength ( $\lambda = 0.71073$  Å). The crystals were glued in paratone to prevent any loss of solvent. An empirical absorption correction was applied using the SADABS program<sup>72</sup> based on the method of Blessing.<sup>73</sup> The structures were solved by direct methods and refined by full-matrix least squares using the SHELX-TL package.<sup>74</sup> Heavier atoms (Mo) were initially located by direct methods. The remaining nonhydrogen atoms were located from Fourier differences and were refined with anisotropic thermal parameters. Positions of the hydrogen atoms belonging to the organic ligand Acac-Bu were calculated and refined isotropically using the gliding mode. Data collection, structure refinement details, and crystallographic data for the single-crystal X-ray diffraction study of 1 can be found in Table S1 (Supporting Information), including the CIF file. CCDC 1854697 also contains the supplementary crystallographic data, which can be obtained free of charge via www.ccdc.cam.ac.uk/data request/cif.

**4.5. Preparation of the Modified Silicon Surfaces.** A piece of p-type  $(1-5~\Omega~{\rm cm}~{\rm resistivity},~{\rm boron\text{-}doped},~250~\mu{\rm m}$  thickness, Siltronix) or n $^+$ -type (0.001  $\Omega~{\rm cm}~{\rm resistivity},~{\rm phosphorus\text{-}doped},~250~\mu{\rm m}$  thickness, Siltronix) Si(100) was sonicated for 10 min successively in acetone, ethanol, and ultrapure 18.2 M $\Omega$  cm water (Veolia Water STI) to degrease it. It was then cleaned with piranha solution (3:1 v/v concentrated  ${\rm H_2SO_4/30\%}~{\rm H_2O_2})$  for 30 min at 100 °C, followed by copious rinsing with ultrapure water to eliminate heavy metals and organic residues.

Caution: The concentrated aqueous H<sub>2</sub>SO<sub>4</sub>/H<sub>2</sub>O<sub>2</sub> (piranha) solution is highly dangerous, particularly in contact with organic materials and should be handled extremely carefully.

For electrochemical measurements, the decontaminated surface was further processed to fabricate the photocathode. An Ohmic contact was established on the backside Si surface with a metal wire by first scrubbing the surface with sand paper and a diamond glass cutter and then applying a droplet of InGa eutectic (99.99%, Alfa Aesar). A layer of silver paste (Electron Microscopy Sciences) was then deposited on the contact. After drying of the silver paste, the metal wire was inserted in a glass capillary, and the electrode area (0.20–0.45 cm²) was defined with an epoxy-based resin (LOCTITE 9460, Henkel) that covered all of the back of the Si surface as well as the silver paste. The assembled electrode was then placed into an oven at 90 °C overnight to cure the resin.

The silicon surface sealed in epoxy was freshly hydrogenated by immersing the electrode into HF 10% wt ( $\sim$ 5.7 M) for 2 min, followed by copious rinsing with ultrapure water and drying under an argon stream.

Caution: Proper precautions must be used when handling hydrogen fluoride. Hydrogen fluoride is extremely corrosive for human tissue, contact resulting in painful, slow-healing burns. Laboratory work with HF should be conducted only in an efficient hood, with the operator wearing a full-face shield and protective clothing.

The cluster solution was prepared at 1 mM in methanol (CHROMASOLV, for high-pressure liquid chromatography,  $\geq$ 99.9%) by dissolving  $[Mo_3S_4(acac-Bu)_3(H_2O)_3]Cl\cdot H_2O$  or  $[(n\text{-}C_6H_{13})_4N]_5AsW_{12}Mo_3S_4(H_2O)_3O_{42}$  followed by sonication for ca. 15 min. The solution was then diluted with methanol to reach the final concentration of 0.5 or 0.1 mM. The freshly hydrogenated Si(100) surface was modified with the  $Mo_3S_4(acac\text{-Bu})$  or  $Mo_3S_4(AsW_{12})$  cluster by drop-casting of the  $[Mo_3S_4(acac\text{-Bu})_3(H_2O)_3]Cl\cdot H_2O$  or  $[(n\text{-}C_6H_{13})_4N]_5AsW_{12}Mo_3S_4(H_2O)_3O_{42}$  solution at 0.1, 0.5, or 1 mM in methanol, yielding cluster loading values of 7.2  $\times$  10 $^{-6}$ , 3.6  $\times$  10 $^{-5}$ , or 7.2  $\times$  10 $^{-5}$  mmol cm $^{-2}$ , respectively. The modified photocathodes were further used for electrochemical measurements after evaporation of the solvent at room temperature for 5–10 min.

**4.6. Electrochemical Measurements.** LSV measurements were performed in a homemade three-neck cell comprising a quartz window in which were inserted a KCl saturated calomel reference electrode (SCE) and a glassy carbon counter electrode (the counter electrode was separated from the rest of the cell by a glass frit). The cell was filled with the electrolytic solution containing either 1.0 mol L $^{-1}$  H<sub>2</sub>SO<sub>4</sub> (pH 0.3) or 0.1 mol L $^{-1}$  KCl + 0.5 mol L $^{-1}$  KHCO<sub>3</sub> (pH 7.3) and was deaerated by bubbling Ar for at least 30 min prior to experiments. The photocathode (hydrogen-terminated p-type Si(100) or freshly covered by the metallic cluster using the protocol described above) was disposed in front of the quartz

window and used as a working electrode. The light was provided by a solar simulator with a fluence of 100 mW cm<sup>-2</sup> (LS0106, LOT Quantum Design) equipped with an AM 1.5G filter. Electrochemical measurements were performed with potentiostat/galvanostat Autolab PGSTAT 302N (Eco Chemie BV) equipped with the GPES and FRA softwares. LSV curves were recorded at 20 mV s<sup>-1</sup>. Unless specified, all of the LSV curves and the reported potentials were intentionally not corrected by the Ohmic drop. The Ohmic drop was determined before each experiment by measuring the impedance of the system at 100 kHz, and it was found to be around 20-100  $\Omega$  depending on the type of the tested photoelectrode and the used electrolytic medium. The geometrical areas of the electrodes used for LSV were in the range 0.20-0.45 cm<sup>2</sup>, and their exact value was measured using the ImageJ software to calculate the current densities. Potentials versus SCE was converted into potential versus RHE using the equation:  $E_{\text{RHE}} = E_{\text{SCE}} + 0.24 + 0.059 \times \text{pH}$ (that was experimentally verified in our operating conditions with a bare platinum electrode). The preparative-scale electrolysis was performed in a Hoffman cell comprising a quartz window and two closed graduated cylinders above the working electrode and the counter electrode. The volumes of H<sub>2</sub> and O<sub>2</sub> gases were measured in situ during the electrolysis; more details can be found in ref 59. For the determination of the Faradaic yield, the theoretical volume of produced H<sub>2</sub>  $(V_{\rm H.}^{\rm T})$  was calculated using the electrical charge used during the electrolysis using the following equation

$$V_{\rm H_2}^{\rm T} = \left(\frac{\rm Q}{2F}\right) \times \left(\frac{RT}{P}\right)$$

with Q being the charge passed during the electrolysis, F being the Faraday constant, R being the gas constant, T being the ambient temperature, and P being the atmospheric pressure.

**4.7.** Additional Instrumentation. XPS measurements were performed with a Mg K $\alpha$  1254 eV X-ray source, using a VSW HA100 photoelectron spectrometer with a hemispherical photoelectron analyzer, working at an energy pass of 20 eV for survey and resolved spectra. The experimental resolution was then 1.0 eV. The binding energy for the main C–C peak has been taken at 285.0 eV as an internal reference level for all measurements. Spectral analysis included a Shirley background subtraction and peak separation using mixed Gaussian—Lorentzian functions.

AFM images were acquired on a NT-MDT NTEGRA microscope in semicontact mode with FM tips (resonance frequency around 60 kHz). The images were treated and analyzed with the open-source Gwyddion software.

### ASSOCIATED CONTENT

## **S** Supporting Information

The Supporting Information is available free of charge on the ACS Publications website at DOI: 10.1021/acsomega.8b01734.

Crystallographic data for  $[Mo_3S_4(acac-Bu)_3(H_2O)_3]Cl-H_2O$ ; spectroscopic characterization of  $Mo_3S_4-POM$ ; supplementary voltammograms and XPS spectra; supplementary AFM images of the modified surfaces; calculation methods of the cluster surface coverage; and determination of the TOF (PDF)

Crystallographic data for the single-crystal X-ray diffraction study of 1 (CIF)

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#### Notes

The authors declare no competing financial interest.

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