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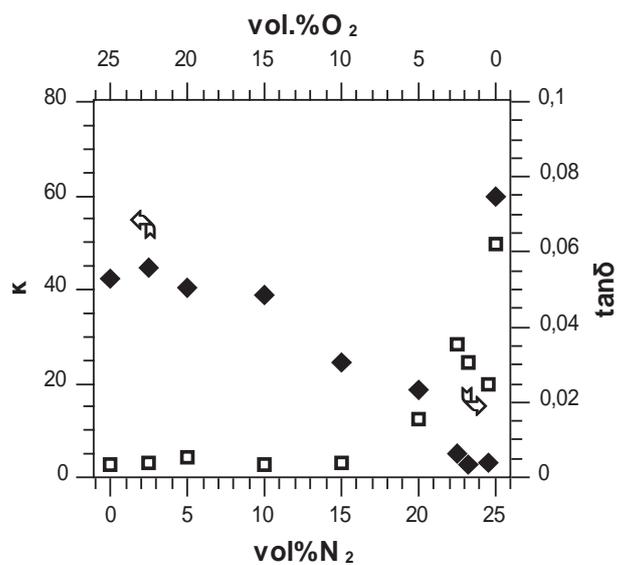
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Graphical Abstract

Evolution of the permittivity κ and loss tangent $\tan\delta$ (@1kHz) of $(\text{Sr}_{0.99}\text{La}_{0.01})_2(\text{Ta}_{0.99}\text{Ti}_{0.01})_2\text{O}_7$ and $(\text{Sr}_{0.99}\text{La}_{0.01})(\text{Ta}_{0.99}\text{Ti}_{0.01})\text{O}_2\text{N}$ thin films as a function of the dinitrogen and dioxygen percentages in the reactive sputtering plasma used for the deposition.

Deposition and dielectric characterization of strontium and tantalum-based oxide and oxynitride perovskite thin films

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Abstract

We have synthesized the composition $x = 0.01$ of the $(\text{Sr}_{1-x}\text{La}_x)_2(\text{Ta}_{1-x}\text{Ti}_x)_2\text{O}_7$ solid solution, mixing the ferroelectric perovskite phases $\text{Sr}_2\text{Ta}_2\text{O}_7$ and $\text{La}_2\text{Ti}_2\text{O}_7$. Related oxide and oxynitride materials have been produced as thin films by magnetron radio frequency sputtering. Reactive sputter deposition was conducted at 750°C under a 75 vol.%(Ar) + 25 vol.%(N₂,O₂) mixture. An oxygen-free plasma leads to the deposition of an oxynitride film $(\text{Sr}_{0.99}\text{La}_{0.01})(\text{Ta}_{0.99}\text{Ti}_{0.01})\text{O}_2\text{N}$, characterized by a band gap $E_g = 2.30$ eV and a preferential (001) epitaxial growth on (001) SrTiO_3 substrate. Its dielectric constant and loss tangent are respectively $\text{Epsilon}' = 60$ (at 1 kHz) and $\text{tanDelta} = 62.5 \cdot 10^{-3}$. In oxygen-rich conditions (vol.%N₂ ≤ 15 %), (110) epitaxial $(\text{Sr}_{0.99}\text{La}_{0.01})_2(\text{Ta}_{0.99}\text{Ti}_{0.01})_2\text{O}_7$ oxides films are deposited, associated to a larger band gap value ($E_g = 4.55$ eV). The oxide films permittivity varies from 45 to 25 (at 1 kHz) in correlation with the decrease in crystalline orientation; measured losses are lower than $5 \cdot 10^{-3}$. For $20 \leq \text{vol.}\% \text{N}_2 \leq 24.55$, the films are poorly crystallized, leading to very low permittivities (minimum $\text{Epsilon}' = 3$). A correlation between the dielectric losses and the presence of an oxynitride phase in the samples is highlighted.

Highlights

- Oxide and oxynitride perovskite films were deposited by magnetron sputtering
- Reactive gas containing dioxygen and dinitrogen was used
- Strontium and tantalum based $(\text{Sr}_{0.99}\text{La}_{0.01})_2(\text{Ta}_{0.99}\text{Ti}_{0.01})_2(\text{O,N})_7$ films were obtained
- The permittivity of the epitaxial oxide film is 42 @1KHz
- The permittivity of the oriented oxynitride film is 60 @1KHz

Keywords

Oxynitride; perovskite; thin films; reactive sputtering; epitaxy; dielectric.

1. Introduction

Among perovskite materials, many original compounds are synthesized from two parent phases, in the form of solid solutions in which the perovskite structure is maintained through cationic substitutions. The aim is to control one or more properties of the material by varying its chemical composition. Ferroelectric perovskite compounds, such as $\text{PbZr}_x\text{Ti}_{1-x}\text{O}_3$ (PZT) [1], $\text{Ba}_x\text{Sr}_{1-x}\text{TiO}_3$ (BST) [2] or $\text{KTa}_x\text{Nb}_{1-x}\text{O}_3$ (KTN) [3], were thus developed with the objective of bringing their Curie temperature (T_C) near room temperature, and therefore, benefit from high values of permittivity, adjustable via an external DC electrical field. The latter property can be used to make frequency agile antennas, i.e. antennas whose resonance frequency can be varied and consequently be used in different standards [4,5]. In miniature antennas, high permittivity perovskite thin films are positioned as the substrate of planar metallic antennas or are integrated in localized components such as capacitances [6]. Agility and miniaturization are two applications targeted for the materials presented in this paper.

$\text{Sr}_2\text{Ta}_2\text{O}_7$ and $\text{La}_2\text{Ti}_2\text{O}_7$, which crystallize in non-centrosymmetric space groups $\text{Cmc}2_1$ and $\text{P}2_1$ respectively, are perovskite ferroelectric compounds: $\text{Sr}_2\text{Ta}_2\text{O}_7$ with $T_C = -107^\circ\text{C}$ [7]

and $\text{La}_2\text{Ti}_2\text{O}_7$ with $T_C = 1466^\circ\text{C}$ [8]. Cross-substitution on the A and B sites of the prototype layered perovskite structure (PLS) $\text{A}_2\text{B}_2\text{O}_7$ would result in the solid solution $(\text{Sr}_{1-x}\text{La}_x)_2(\text{Ta}_{1-x}\text{Ti}_x)_2\text{O}_7$. Here, we have interpolated the results of Nanamatsu *et al.* on single crystals and ceramics [7] and determined the stoichiometry $x \sim 0.01$ giving a value of T_C close to room temperature. The present paper deals with the synthesis of the $(\text{Sr}_{0.99}\text{La}_{0.01})_2(\text{Ta}_{0.99}\text{Ti}_{0.01})_2\text{O}_7$ (SLTTO) compound as thin films deposited by radio-frequency (rf) reactive magnetron sputtering. Moreover, we were interested in the deposition of the oxynitride $(\text{Sr}_{0.99}\text{La}_{0.01})(\text{Ta}_{0.99}\text{Ti}_{0.01})\text{O}_2\text{N}$ (SLTTON) compound, where substitution occurs on the anionic sublattice because of similar values of ionic radii and electronegativity for oxygen and nitrogen ($r_{\text{O}^{2-}} = 1.40 \text{ \AA}$; $r_{\text{N}^{3-}} = 1.50 \text{ \AA}$; $\chi_{\text{O}} = 3.44$; $\chi_{\text{N}} = 3.04$). Oxynitride perovskites are currently studied because of their original properties compared to their oxide parents, in particular an absorption in the visible domain leading to their use as photocatalyst in the water-splitting reactions [9,10], and very high permittivities [11-14], with a potential use in agile and/or miniaturized antennas. Some reports concern these oxynitride compounds in the form of thin films [15-17].

In recent years, our research has been concerned with the deposition of oxide and oxynitride perovskite films for a use in electronic and telecommunication devices such as antennas. We have recently reported on SLTTO and SLTTON films deposited by reactive rf magnetron sputtering using $\text{Ar} + \text{O}_2$ or $\text{Ar} + \text{N}_2$ plasma, respectively [18]; the resulting films were epitaxially grown on (001) MgO substrates. The objective of the present study is to determine whether a mixture of the SLTTO and SLTTON compounds could be obtained using an original reactive plasma, containing Ar and *both* O_2 and N_2 . Considering that free energies of formation of oxide compounds are lower than those of oxynitrides [19], this sort of dual-reactive medium has rarely been used up to now for the deposition of films using an oxide precursor [20]. Usually, on powders, an oxide precursor exposed to an oxidizing and

nitriding atmosphere leads to a stable oxide, whereas an (oxy)-nitride or metallic precursor under an oxidizing and nitriding atmosphere (O_2/N_2 , for example [21]) can lead to an original oxynitride material. Here, the synthesis of the compounds as thin films, using an energetic deposition process such as sputtering deposition under dual-reactive plasma, allows us to consider that this (oxy)nitride synthesis route could be tested. Furthermore, as detailed hereafter, the mixing of SLTTO and SLTTON materials was intended in order to reduce the dielectric loss of films.

Given the very small amount (1%) of the $La_2Ti_2O_7$ material in the synthesized compounds, we will discuss our results in reference to closest compounds, $Sr_2Ta_2O_7$ and $SrTaO_2N$. For $Sr_2Ta_2O_7$, permittivities from 20 to 95 have been measured on thin films [22-24]; the dispersion of values can be explained by the different methods of film deposition (atomic vapor deposition, sol-gel), crystallization (amorphous, polycrystalline) and thickness of films (from 15 to 250 nm). The available data on $SrTaO_2N$ dielectric performance are scarce; on bulk samples, permittivities close to 3000 have been published by Kim *et al.* [11] and Hinuma *et al.* [12], while Masubuchi *et al.* reported very recently on permittivities in the range 60 -200 for ceramics obtained by high pressure densification [13]. On thin films, Oka *et al.* reported on $SrTaO_2N$ films deposited by nitrogen plasma enhanced-laser ablation [25]. These films were epitaxially grown on conducting niobium-doped (001) $SrTiO_3$ substrates and reveal permittivity around 2000 at 1 kHz at room temperature, with dielectric losses $\tan\delta$ close to $35 \cdot 10^{-3}$. PFM measurements also pointed out the piezoelectric and ferroelectric behavior of films; Oka *et al.* proposed the material to be constituted of a relaxor ferroelectric-type matrix surrounding classical ferroelectric-type nano-domains. Besides these studies, we have also published on oxynitride $LaTiO_2N$ (LTON) films; their permittivities are dependent on their structure and microstructure ($\kappa_{LTON} \sim 375 - 1220$ [26,27]), but are always higher than the $La_2Ti_2O_7$ (LTO) oxide ($\kappa_{LTO} \sim 60$ [28,29]). The drawback lies in higher losses, with $\tan\delta$

from 0.06 to 1 for LTON thin films. For this reason, the present study does not only concern the synthesis of SLTTO oxide and SLTTON oxynitride, but the possibility of producing mixtures of phases, or even forming a $(\text{Sr}_{0.99}\text{La}_{0.01})_2(\text{Ta}_{0.99}\text{Ti}_{0.01})_2(\text{O,N})_7$ (SLTT) solid solution and thereby modulate the nitrogen content in films and so, reduce the dielectric losses. For that purpose, we conducted a series of deposits using an original reactive sputtering gas composed of 75 vol.% of argon and 25 vol.% of dioxygen *and* dinitrogen. This paper deals with the deposition of such samples; it presents the structural and microstructural characterization of the oxide and oxynitride deposited films and gives the first results of their dielectric characterization at room temperature in the low frequency range. The study concerning the ferroelectric properties, in particular their temperature dependence, will concern other samples and will be presented in a separate paper.

2. Materials and Methods

Film deposition is carried out using rf reactive magnetron sputtering in a Plassys MP450S reactor using a homemade $(\text{Sr}_{0.99}\text{La}_{0.01})_2(\text{Ta}_{0.99}\text{Ti}_{0.01})_2\text{O}_7$ target. It is obtained by uniaxial compaction of SLTTO powder synthesized by solid state chemical route from stoichiometric amounts of SrCO_3 , Ta_2O_5 , La_2O_3 and TiO_2 heated under air at 1400°C during 30 h. X-ray diffraction (XRD) analysis confirms the crystallization of the powder according to a $\text{Sr}_2\text{Ta}_2\text{O}_7$ analogue (see Supplementary Data 1).

A series of films is deposited using the same sputtering conditions, except the gas mixture. The plasma mixture always contains 75vol.% of argon plus 25vol.% dedicated to the reactive part, i.e. x vol.% N_2 + y vol.% O_2 , with $x + y = 25$; deposited films will be labeled as SLTT- $x\text{N}_2$ - $y\text{O}_2$ (for example, SLTT-2.5 N_2 -22.5 O_2 has been deposited under a mixture of vol.% N_2 = 2.5 and vol.% O_2 = 22.5). The other deposition parameters are the following: total pressure $p_T = 40$ mTorr; rf power $P_{\text{rf}} = 90$ W (2.04 W/cm²); substrate temperature $T_S = 750^\circ\text{C}$

and target-substrate distance $d = 5$ cm. Before deposition, a pre-sputtering of the target is performed during 100 min under the same gas mixture than deposition. The films are cooled under the same gas mixture than deposition; no post-annealing is performed.

Two kinds of single-crystalline substrates are used: (001) MgO and (001) Nb:SrTiO₃. The MgO substrate is used to determine the band-gap of the film from UV-visible spectra, while conducting Nb(1.4%):SrTiO₃ is preferred to make Metal - Insulating - Metal (MIM) structures for the dielectric measurements.

During deposition, the plasma is monitored by Optical Emission Spectroscopy (OES) using an Ocean Optics HR4000 monochromator, with the position of the optical fiber kept constant from one experiment to another. The emissions of Ar and N₂ are followed at 750.4 nm ($2p_1 \rightarrow 1s_2$) [30] and 357.0 nm ($C^3\Pi_u \rightarrow B^3\Pi_g$) [31] respectively. In order to decrease the effect of the variation of the electron temperature, the N₂ band intensity is divided by the one of the Ar line.

Thickness of layers is measured by Scanning Electron Microscope (SEM) cross-sectional view, using a JEOL 5440 SEM; error on thickness value is estimated as being lower than 20 nm. The chemical composition of samples is determined by semi-quantitative Energy Dispersive Spectrometry (EDS) in a JEOL5440 SEM operating at 20 kV; lower limit of detection is around 2 at.%. UV-Visible transmittance spectra are collected by a Perkin-Elmer Lambda 20 spectrometer operating in the range [200-1100] nm. Assuming direct transitions, band-gap values are obtained using the relation:

$$(\alpha h\nu)^2 = (h\nu - E_g)$$

where $h\nu$ is the photon energy. Hence plotting $(\alpha h\nu)^2$ as a function of $h\nu$, the extrapolation of the linear region at zero gives the value of E_g . The error is estimated as 0.05 eV.

X ray diffractograms are collected using a Seifert XRD 3003 diffractometer with a copper anode. Conventional $\theta - 2\theta$ and ϕ scans are recorded. In order to quantify the thin

film orientation, the Lotgering factor is used. This factor compares the orientation of a thin film versus randomly oriented material (i.e. powder) and is able to vary from 0 for a non-oriented layer to 1 for a total oriented layer. The Lotgering factor is calculated using the relations [32]:

$$F_{hkl} = \frac{P_{hkl} - P_{0(hkl)}}{1 - P_{0(hkl)}}$$

$$\text{with } P_{(hkl)} = \frac{\sum_i I_{(hkl)}}{\sum_i I_{(h'k'l')}} \text{ and } P_{0(hkl)} = \frac{\sum_i I_{0(hkl)}}{\sum_i I_{0(h'k'l')}}$$

where (hkl) is used for the studied orientation of the layer, (h'k'l') are used for the other plane families, I(hkl) is the intensity of the (hkl) indexed peak and the subscript 0 is relative to the powder reference diffractogram (JCPDS data).

Low frequency dielectric properties are measured at room temperature by a LCR meter (LCR-819 GWInstek) on Metal/Insulator/Metal (MIM) structures. The conducting substrate (Nb:SrTiO₃) is used as bottom electrode, whereas 200 nm sputtered silver serves as the top electrode. Measurements are also done on a dense (Sr_{0.99}La_{0.01})₂(Ta_{0.99}Ti_{0.01})₂O₇ ceramic, obtained by uniaxial compaction of oxide powder and sintering at 1540 °C under air during 2 h, with 200 nm silver sputtered on both sides of the pellet to achieve a MIM structure.

3. Results

Table 1 gathers different characteristics of the as-prepared films, such as their thickness, band gap and Lotgering factor, as well as the reactive N₂/O₂ mixtures used for the depositions.

3.1 Reactive sputtering and deposition rate

Before the series of deposition, an OES study of the plasma was carried out to investigate the poisoning of the oxide target during reactive sputtering with dioxygen and dinitrogen. Figure 1 presents the evolution of the intensity ratio of the emission peak of

dinitrogen vs. argon (I_{N_2} / I_{Ar}) as a function of sputtering time for a target previously pre-sputtered under dioxygen (vol.% $O_2 = 25$ / vol.% $N_2 = 0$) and undergoing a sputtering under dinitrogen (vol.% $O_2 = 0$ / vol.% $N_2 = 25$). The onset point corresponds to the changeover from an Ar/ O_2 plasma to a N_2 -containing plasma. A continuous increase of I_{N_2} / I_{Ar} is observed during the first 60 min of sputtering, leading to a plateau. Such feature is characteristic of a nitridation of the target: during the first hour of the process, injected N_2 is partially consumed in a progressive nitridation of the target surface (and the walls of the deposition chamber), then, a balanced regime is reached. That leads us to choose a sufficient pre-sputtering time of 100 minutes. Furthermore, Figure 1 also demonstrates that the chosen emission of N_2 at 357.0 nm is a valuable parameter to follow the evolution of the reactive sputtering.

Concerning the series of deposition, Figure 2 shows the evolution of the deposition rate as a function of the composition of the deposition plasma (we remind that pre sputtering is performed under the same atmosphere as the deposition). We observe a gradual increase of the deposition rate when the percentage of dinitrogen in the sputtering gas increases, with a greater increase around vol.% $N_2 = 20$. Figure 2 also draws the evolution of the intensity ratio of the emission peak of dinitrogen vs. argon (measured by OES at the end of the pre-sputtering, just before the deposition): the two curves show similar trends, with an inflection point between 15 and 20 vol.% N_2 . These behaviors are relative to the poisoning of the oxide target when sputtered by dinitrogen and will be explained later in the article based upon the results of deposited films.

3.2 Optical characterization of films

Transmittance measurements show that films synthesized with dinitrogen ratio between 0 and 24.55 % are transparent in the visible region and lead to band gaps around 4.55 eV, as seen in Table 1 and Supplementary Data 2. The film deposited without oxygen is

orange and has a band-gap of 2.30 eV. These values are in agreement with literature about $\text{Sr}_2\text{Ta}_2\text{O}_7$ and SrTaO_2N which leads to band gaps around 4.50 - 4.60 eV [33-34] and 2.1 - 2.27 eV [35,25], respectively. Furthermore, it can be noted that there is no progressive decrease of the band gap with the increase of dinitrogen ratio in the gas mixture. The sharp drop in the band gap is only obtained for the SLTT-25 N_2 -0 O_2 sample, for which there is no oxygen in the deposition gas mixture. Such decrease is attributed to substitution of oxygen for nitrogen in the film, thus producing an oxynitride compound. As oxygen is more electronegative than nitrogen, $2p(\text{N})$ states will lie above $2p(\text{O})$ states at the top of the valence band, leading to the observed narrowing of the band-gap for the SLTT-25 N_2 -0 O_2 sample. EDS characterization confirms the presence of nitrogen only in the SLTT-25 N_2 -0 O_2 sample.

3.3 Crystallization of films

θ - 2θ X-ray diffractograms of deposited thin films are shown in Figure 3. Films synthesized with $0 \leq \text{vol.}\% \text{N}_2 \leq 10$ are clearly indexed as the oxide $\text{Sr}_2\text{Ta}_2\text{O}_7$ (JCPDS 72-0921) ; a systematic angular deviation with reference is observed. A preferred $\langle 110 \rangle$ orientation is visible, quantified by Lotgering factors equal or very close to 1 as shown in Table 1. This preferred orientation has been already reported in the case of $\text{Sr}_2\text{Ta}_2\text{O}_7$ nanosheets growth [35]. One can detect a limit (vol. $\text{N}_2 = 15\%$) beyond which the crystallinity of samples drastically decreases as reflected by the decline in the intensity of the diffracted peaks as well as falling values of the Lotgering factors. The SLTT-15 N_2 -10 O_2 film is still referred as the oxide with the appearance of the (151) peak, but, above, the diffracted peaks are better indexed as the oxynitride SrTaO_2N compound, with a slight (001) oriented growth. The last sample deposited with $\text{vol.}\% \text{N}_2 = 25$ is highly crystallized, indexed as SrTaO_2N with a strong preferred c-axis orientation as proved by its Lotgering factor $F_L =$

0.995. No impurity phases are evidenced, as SrTa_2O_6 and $\text{SrTa}_4\text{O}_{11}$ detected in the powdered target.

The above results are emphasized in Figure 4 drawing the evolution of the inter-reticular distances (d_{hkl}) of orientation of films as a function of the percentages of N_2 and O_2 in the sputtering plasma. The two sets of synthesized films are highlighted: for $\text{vol}\% \text{N}_2 \leq 15$, d_{hkl} are assigned to d_{110} of the oxide compound and are close to a value of 3.985 Å, whereas above, d_{hkl} assigned to d_{002} of the oxynitride are close to 4.070 Å. Values for films are always higher than $\text{Sr}_2\text{Ta}_2\text{O}_7$ and SrTaO_2N references on powder. When comparing the distances of the present films deposited on SrTiO_3 (001) substrates to those of the twin films deposited on MgO (001) (see Supplementary Data 3), nearly identical d_{hkl} are obtained despite very different cubic cell parameters for the substrates ($c_{\text{STO}} = 3.905$ Å; $c_{\text{MgO}} = 4.211$ Å). This implies that the films are no longer strained on these substrates due to their high thickness (≥ 665 nm); the experimental d_{hkl} distances are thus representative of the deposited oxide SLTTO and oxynitride SLTTON thin film materials.

Films deposited at $\text{vol}\% \text{N}_2 = 0$ and 25 exhibit a strong preferred orientation, hence epitaxial growth can be expected. φ scans, recorded at $\chi = 45^\circ$ on the (151), (112) and (101) poles of the $(\text{Sr}_{0.99}\text{La}_{0.01})_2(\text{Ta}_{0.99}\text{Ti}_{0.01})_2\text{O}_7$, $(\text{Sr}_{0.99}\text{La}_{0.01})(\text{Ta}_{0.99}\text{Ti}_{0.01})\text{O}_2\text{N}$ and SrTiO_3 compounds respectively, are shown in Figure 5. One can see four peaks distanced by 90° . This supports the assumption that the studied compounds can be seen as pseudo-cubic structures, like most of perovskites, because their lattice constant c is close to $a\sqrt{2}$. The same positions for the film and substrate peaks show an alignment of the film layers with those of substrates in the $\chi = 45^\circ$ plane. Along with the orientation of films in the $\chi = 0^\circ$ plane, as observed in Figure 3, this demonstrates the epitaxial growth of the oxide SLTT-0 N_2 -25 O_2 and oxynitride SLTT-25 N_2 -0 O_2 samples on SrTiO_3 substrate. Nevertheless, the epitaxy for the oxynitride is not complete because peaks other than (00 l) family ones exist in its θ -2 θ

diffractogram. Furthermore, in ϕ -scan, the full width at half maximum of the oxynitride film peaks is $\Delta\phi = 6.25^\circ$, indicating a substantial dispersion of the crystallographic alignment. This is explained by the relatively high mismatch ($m = + 4.2 \%$) between the SLTTON film and the SrTiO_3 substrate, calculated using the pseudo-cubic parameter of the SLTT-25 N_2 -0 O_2 oxynitride film ($c_{\text{PC-ON}} = 4.069 \text{ \AA}$). For this sample, it is therefore more appropriate to refer to a three-dimensional preferred epitaxy. Furthermore, no lattice tetragonal distortion of the pseudo-cubic cell is revealed here ($c_{\text{PC-ON}} / a_{\text{PC-ON}} = 0.996$), as the one emphasized by Oka *et al.* on epitaxial SrTaO_2N films grown on SrTiO_3 substrates [25]. By combining informations from Figures 3 and 5, the SLTT-0 N_2 -25 O_2 sample can be considered as fully epitaxially grown. This can be explained by the relatively low mismatch ($m = + 1.9 \%$) between the SLTTO oxide film, considered as pseudo-cubic ($c_{\text{PC-O}} = 3.978 \text{ \AA}$), and the SrTiO_3 substrate. Nevertheless, in the plane of the sample, the dispersion of the crystallographic alignment between the (151) film plane and (101) substrate plane is high ($\Delta\phi = 8.15^\circ$), so the epitaxy of the oxide film on the SrTiO_3 substrate is qualified as moderate.

3.4 Dielectric characterization of ceramic and films

Dielectric properties are measured at 1 kHz. This frequency is chosen because, beyond this value, dielectric losses of the oxide materials are too low ($\tan\delta < 10^{-4}$) and cannot be measured by our experimental setup.

Dielectric measurements performed on the oxide SLTTO ceramic give the following values: permittivity $\kappa_{\text{ceramic}} = 54$ and dielectric losses $\tan\delta_{\text{ceramic}} < 10^{-4}$. This emphasizes moderate permittivity and very low losses.

For the films, the evolution of the permittivity κ and loss tangent $\tan\delta$ as a function of the dinitrogen and dioxygen percentages in the sputtering plasma is depicted in Figure 6; values are detailed in Table 2.

The permittivity values are in the range 39 - 45 for the films deposited with $\text{vol.\%N}_2 \leq 10$. Above this threshold N_2 amount, the films permittivity significantly decreases to a minimum of $\kappa = 3$ for $\text{vol.\%N}_2 = 23.25$. On the other hand, the film deposited at $\text{vol.\%N}_2 = 25$ has the highest permittivity ($\kappa = 60$).

Concerning the losses, the lowest values can be obtained for $\text{vol.\%N}_2 \leq 15$ ($\tan\delta \sim 3.5 - 5.0 \cdot 10^{-3}$); above losses increase up to a $\tan\delta$ of $62.5 \cdot 10^{-3}$ for $\text{vol.\%N}_2 = 25$.

4. Discussion

All the results of the present study point out a significant change in the reactive sputtering mechanism at a plasma dinitrogen percentage at 15-20 vol.%.

The curves presented in Figure 2 both show an inflection point around 15 vol.% N_2 . Up to that threshold, and despite a 100 minutes pre-sputtering time, the O_2/N_2 ratio in the reactive gas leads to a partially nitrided target surface. That way, the depositions for $\text{vol.\%N}_2 \leq 15$ produce oxide films as underlined by low deposition rates, transparency in visible region, oxide signature in XRD and very low dielectric loss. Plasmas with higher N_2 percentages lead to films with an intermediate structure, except for the deposition without dioxygen for which a pure oxynitride film is obtained, with the proper band gap (2.30 eV) and crystallization, and greater dielectric loss. High deposition rates obtained under N_2 -rich plasma are coherent with a nitrogen enrichment of the target surface during the deposition process ; as a nitrided target shows higher sputtering and secondary electron emitting yields than an oxide target [36,37] deposition rates are here logically higher for oxynitride than for oxide films.

Films deposited with $20 \leq \text{vol.\%N}_2 \leq 24.55$ have been characterized by an absorption in the U-V region (with band gaps around 4.55 eV), no nitrogen detected by EDS, but an oxynitride XRD signature and dielectric losses higher than those of oxide films. These complex films could be considered as mixtures of a small amount of low crystallized

$(\text{Sr}_{0.99}\text{La}_{0.01})_2(\text{Ta}_{0.99}\text{Ti}_{0.01})_2\text{O}_2\text{N}$ oxynitride phase and a high proportion of amorphous $(\text{Sr}_{0.99}\text{La}_{0.01})_2(\text{Ta}_{0.99}\text{Ti}_{0.01})_2\text{O}_7$ oxide phase conferring them a transparency in the visible range. The presence of the oxynitride phase will explain their increased dielectric losses, as already mentioned for oxynitride ceramics compared to oxide ones [11]. Given the epitaxial “cube on cube” growth for the pure oxide and pure oxynitride films, the low crystallization of these intermediate films could be explained by a difference in the perovskite structure of the compounds. They both belong to the family of tantalum (and titanium) layered perovskite structures described by Lichentberg as $\text{A}_n\text{B}_n\text{X}_{3n+2}$ (here, A = Sr, La; B = Ta, Ti and X = O, N) [38]. The $(\text{Sr}_{0.99}\text{La}_{0.01})_2(\text{Ta}_{0.99}\text{Ti}_{0.01})_2\text{O}_7$ oxide is a $n = 4$ member, made by the stacking of four $(\text{Ta,Ti})\text{O}_6$ octahedra separated by interlayers containing Sr (and La) and O atoms. The $(\text{Sr}_{0.99}\text{La}_{0.01})_2(\text{Ta}_{0.99}\text{Ti}_{0.01})_2\text{O}_2\text{N}$ oxynitride is relative to $n = \infty$, with infinite stacking of octahedra in the c direction. These two stacking modes are so different that it could be difficult to stabilize the two structures in one material. In the present study, it was not possible, either in the form of a well-crystallized mixture of phases, nor as a solid solution; can't the amount of nitrogen in films be controlled by this way.

Concerning the dielectric performance of the oxide materials, results obtained on the SLTTO ceramic can be compared to those of the oxide SLTT-0_{N2}-25_{O2} film. The film permittivity is slightly lower ($\kappa = 42$) than that of the ceramic material ($\kappa = 56$) and the losses are at least higher by one order of magnitude. These values are in the range of those measured on $\text{Sr}_2\text{Ta}_2\text{O}_7$ films ($\kappa = 20 - 95$ [22,24]). The performance gap between ceramic and film is also in agreement with literature reports. Grain size is the key parameter that could affect the dielectric characteristics of films compared with ceramics [39], because smaller grains in films induce more grain-boundary dead layers leading to a low permittivity. Here, as measured by SEM (Figure 7), the grain size is estimated at $\phi \sim 100$ nm for the film and is in the range 1 - 10 μm for the ceramic.

For the series of films, characterizations show that the quality of crystallization is correlated with permittivity values: high values are obtained for the fully or preferentially epitaxial films (SLTT-0_{N₂}-25_{O₂} and SLTT-25_{N₂}-0_{O₂}), whereas low κ values are obtained for the poor crystallized intermediate films.

For the oxynitride SLTT-25_{N₂}-0_{O₂} film, the permittivity value (60) is close to the values reported on SrTaO₂N ceramics obtained by high pressing sintering [13] which range from 60 to 200 (@100kHz) with losses $\tan\delta > 0.1$. At the contrary, our results are very far from those published on a SrTaO₂N film [25], with a permittivity value above 2000 at 1 kHz (RT). This film, with thickness 288 nm, shows a tetragonal distortion, leading to a model based on a trans-type N ordering in small domains, thus exhibiting a classical ferroelectric behavior and high permittivity. In our study, the oxynitride film is perfectly relaxed because of its high thickness which can hide the high permittivity contribution of the first strained layers. Future work on the influence of thickness on the dielectric characteristics of oxynitride films will highlight this point.

Finally, our study shows that the dielectric characterization, beyond giving values to evaluate the integration of films in electronic and telecommunication devices, act as a complementary characterization of compounds containing nitrogen and can be as (more) sensitive than other methods.

5. Conclusion

(Sr_{0.99}La_{0.01})₂(Ta_{0.99}Ti_{0.01})₂O₇ oxide and (Sr_{0.99}La_{0.01})₂(Ta_{0.99}Ti_{0.01})₂O₂N oxynitride are related to the solid solution between the ferroelectric Sr₂Ta₂O₇ and La₂Ti₂O₇ perovskite compounds. They were synthesized as thin films by magnetron rf sputtering using an oxide target under a reactive gas composed of dioxygen and dinitrogen with percentages varying inversely from 0 to 25. Different kinds of films were obtained. For $0 \leq \text{vol.\%N}_2 \leq 15$, the

films are assigned to the $(\text{Sr}_{0.99}\text{La}_{0.01})_2(\text{Ta}_{0.99}\text{Ti}_{0.01})_2\text{O}_7$ oxide isostructural to the $\text{Sr}_2\text{Ta}_2\text{O}_7$ material, with a band-gap close to 4.55 eV. The films are epitaxially grown or (110) oriented on (001) SrTiO_3 substrates. Their permittivities are in the range of $\kappa = 45 - 25$ @1 kHz with values decreasing when the percentage of dinitrogen in plasma grows; the dielectric losses are inferior to $5 \cdot 10^{-3}$. For $\text{vol.}\% \text{N}_2 = 25$, the film corresponds to the $(\text{Sr}_{0.99}\text{La}_{0.01})(\text{Ta}_{0.99}\text{Ti}_{0.01})\text{O}_2\text{N}$ oxynitride, isostructural to SrTaO_2N , with a band gap of 2.30 eV. It is preferentially 3D epitaxially grown on (001) SrTiO_3 substrate. Its permittivity is the highest of the series of depositions ($\kappa = 60$ @1 kHz), as well as its dielectric losses ($62.5 \cdot 10^{-3}$). In between, that is for $20 \leq \text{vol.}\% \text{N}_2 \leq 24.55$, the films are poorly crystallized with an oxynitride contribution and could correspond to mixtures of the oxide and the oxynitride phases. The permittivities are low ($\kappa = 18.5 - 2.5$) and their increasing dielectric losses could be explained by the oxynitride contribution. Finally, SLTTO oxide films, with very small $\tan\delta$ values and relatively high permittivities, could be used as low dielectric loss material in electronic and telecommunication devices.

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Table 1.

List of films deposited by reactive rf sputtering from a $(\text{Sr}_{0.99}, \text{La}_{0.01})_2(\text{Ta}_{0.99}, \text{Ti}_{0.01})_2\text{O}_7$ target on MgO and Nb-SrTiO₃ substrates, along with their thickness, band-gap and Lotgering factor. The samples are labelled as SLTT- $x\text{N}_2$ - $y\text{O}_2$, with x the percentage of N₂ and y the percentage of O₂ in the sputtering plasma.

Sample	%vol.N ₂	%vol.O ₂	Thickness (nm)	Band gap (eV)	Lotgering Factor F _L
SLTT-0N ₂ -25O ₂	0	25	775	4.60	1
SLTT-2.5N ₂ -22.5O ₂	2.5	22.5	870	4.45	1
SLTT-5N ₂ -20O ₂	5	20	700	4.65	1
SLTT-10N ₂ -15O ₂	10	15	830	4.40	0.995
SLTT-15N ₂ -10O ₂	15	10	665	4.45	0.635
SLTT-20N ₂ -5O ₂	20	5	775	4.60	0.805
SLTT-22.5N ₂ -2.5O ₂	22.5	2.5	735	4.70	0.875
SLTT-23.25N ₂ -1.75O ₂	23.25	1.75	770	4.60	0.745
SLTT-24.55N ₂ -0.45O ₂	24.55	0.45	885	4.65	0.720
SLTT-25N ₂ -0O ₂	25	0	1120	2.30	0.995

Table 2.

Permittivity κ and loss tangent $\tan\delta$ at 1 kHz of the oxide ceramic and SLTT films labelled as SLTT- xN_2 - yO_2 , with x the percentage of N_2 and y the percentage of O_2 in the sputtering plasma.

Sample	κ (@1kHz)	$\tan\delta$ (@1kHz)
SLTT-0 N_2 -25 O_2	42	$3.5 \cdot 10^{-3}$
SLTT-2.5 N_2 -22.5 O_2	45	$4.0 \cdot 10^{-3}$
SLTT-5 N_2 -20 O_2	41	$5.0 \cdot 10^{-3}$
SLTT-10 N_2 -15 O_2	39	$3.5 \cdot 10^{-3}$
SLTT-15 N_2 -10 O_2	25	$4.0 \cdot 10^{-3}$
SLTT-20 N_2 -5 O_2	19	$15.5 \cdot 10^{-3}$
SLTT-22.5 N_2 -2.5 O_2	5	$35.0 \cdot 10^{-3}$
SLTT-23.25 N_2 -1.75 O_2	3	$30.5 \cdot 10^{-3}$
SLTT-24.55 N_2 -0.45 O_2	3	$24.5 \cdot 10^{-3}$
SLTT-25 N_2 -0 O_2	60	$62.5 \cdot 10^{-3}$
SLTTO ceramic	54	$< 10^{-4}$

Figure Captions

Figure 1. Evolution of the intensity ratio of the emission peak of dinitrogen vs. argon (I_{N_2} / I_{Ar}) recorded by OES as a function of reactive sputtering time, with a plasma containing 75 vol.%Ar and 25 vol.% N_2 and a target initially oxidized (pre-sputtered with 75vol.%Ar and 25 vol.% O_2).

Figure 2. Evolution of the deposition rate of SLTT films and the ratio of emitted intensities recorded by OES as a function of dinitrogen and dioxygen percentages in the reactive sputtering plasma.

Figure 3. $\theta - 2\theta$ X-ray diffractograms of SLTT thin films deposited on Nb:SrTiO₃(001) (STO) substrates labelled as SLTT- xN_2 - yO_2 , with x the percentage of N_2 and y the percentage of O_2 in the sputtering plasma (dotted lines underline some peaks indexed as an oxynitride; solid lines underline some peaks indexed as an oxide).

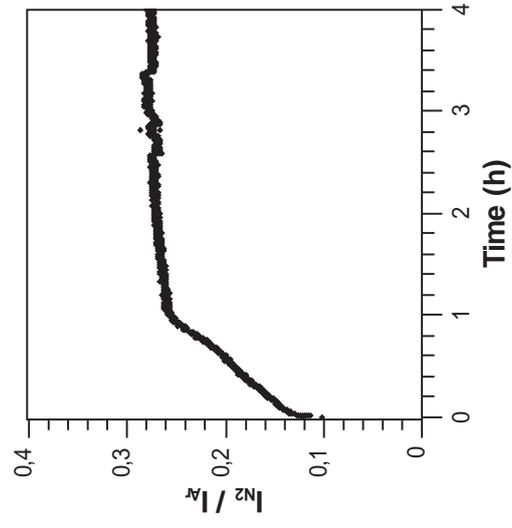
Figure 4. Calculated inter-reticular d_{hkl} distances of orientation of SLTT films as a function of the dinitrogen and dioxygen percentages in the reactive sputtering plasma. Dotted line draws tabulated values of SrTaO₂N (d_{002}) and solid line tabulated values of Sr₂Ta₂O₇ (d_{110}). The error bar is less than the symbol size.

Figure 5. X-ray diffraction φ scans at $\chi = 45^\circ$ of: (a) the film deposited without dinitrogen (SLTT-0 N_2 -25 O_2 , pole (151) and its SrTiO₃ substrate (pole (101)), (b) the film deposited without dioxygen (SLTT-25 N_2 -0 O_2 , pole (112) and its SrTiO₃ substrate (pole (101))).

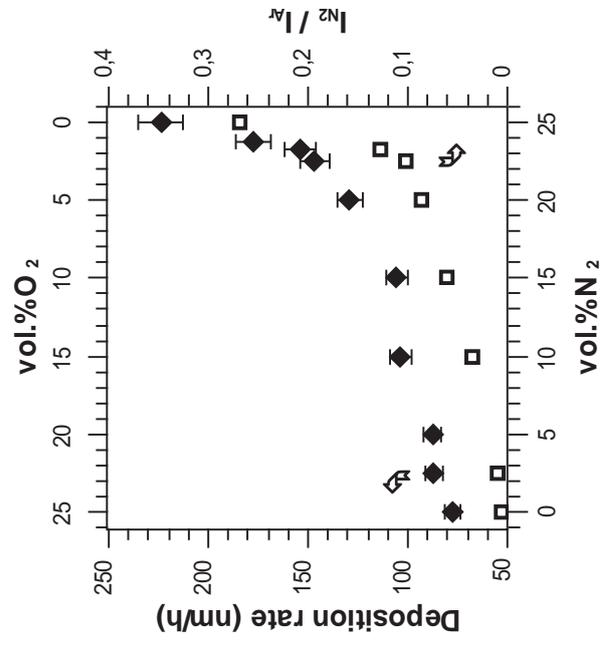
Figure 6. Evolution of the permittivity κ and loss tangent $\tan\delta$ of SLTT films as a function of the dinitrogen and dioxygen percentages in the reactive sputtering plasma (the error bar is the symbol size).

Figure 7. Surface morphology seen by scanning electron microscopy of the (Sr_{0.99}La_{0.01})₂(Ta_{0.99}Ti_{0.01})₂O₇ material as (a) thin film (SLTT-0 N_2 -25 O_2 sample) and (b) ceramic (the magnifications are different because of dissimilar grain sizes).

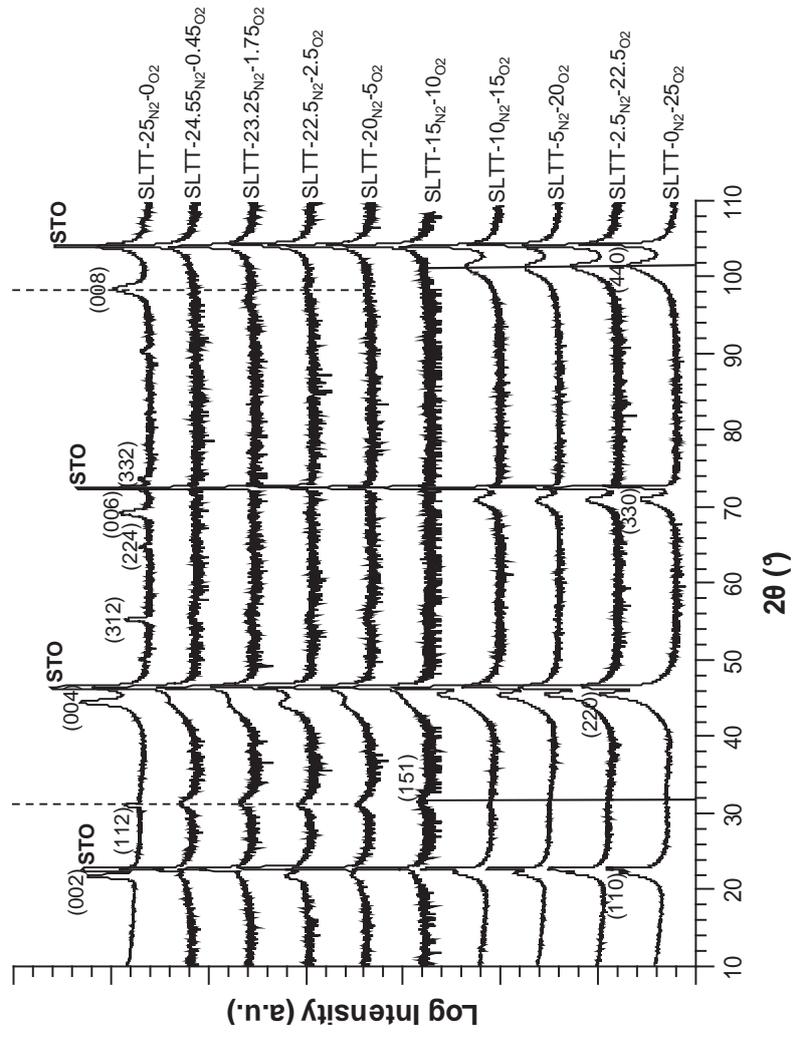
Jacq_Figure 1_revised



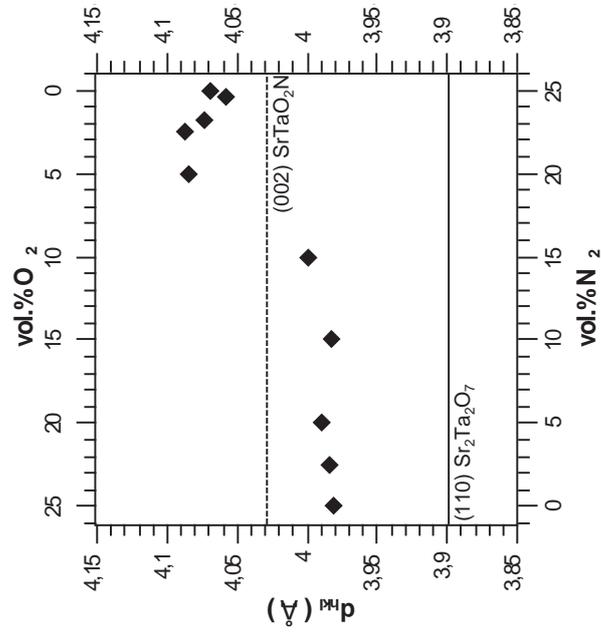
Jacq_Figure 2_revised



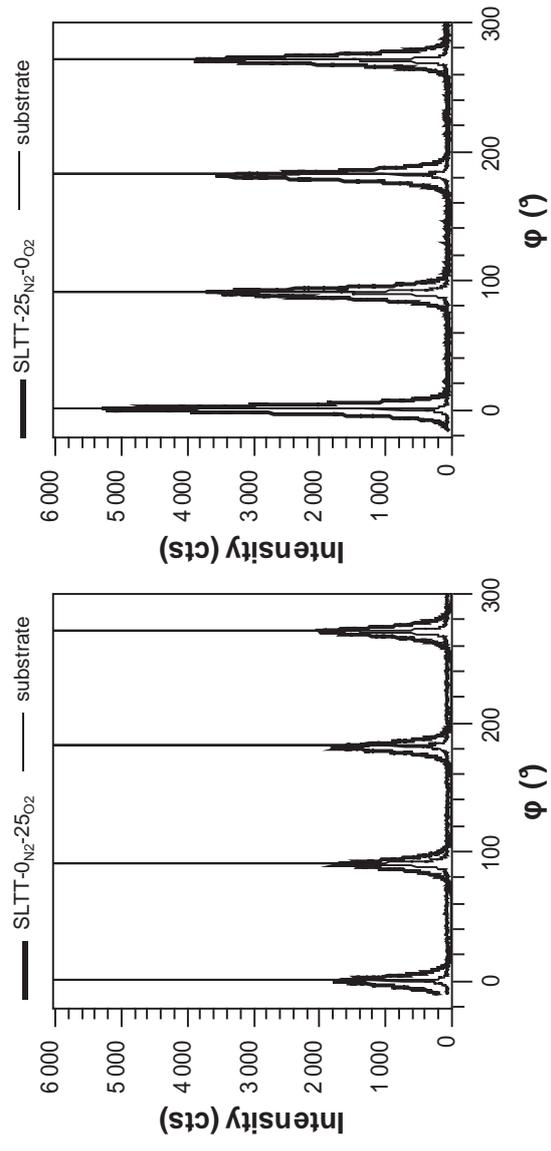
Jacq_Figure 3_revised



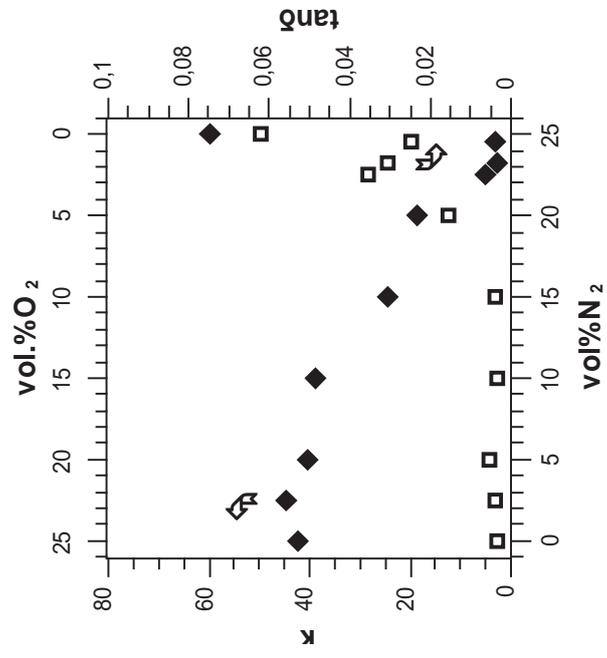
Jacq_Figure 4_revised



Jacq_Figure 5_revised



Jacq_Figure 6_revised



Jacq_Figure 7_revised

